

# Introduction to high energy resolution X-ray spectroscopies

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Part of the course: Optical Spectroscopy in Materials Science

BME Physics

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# 1. Introduction

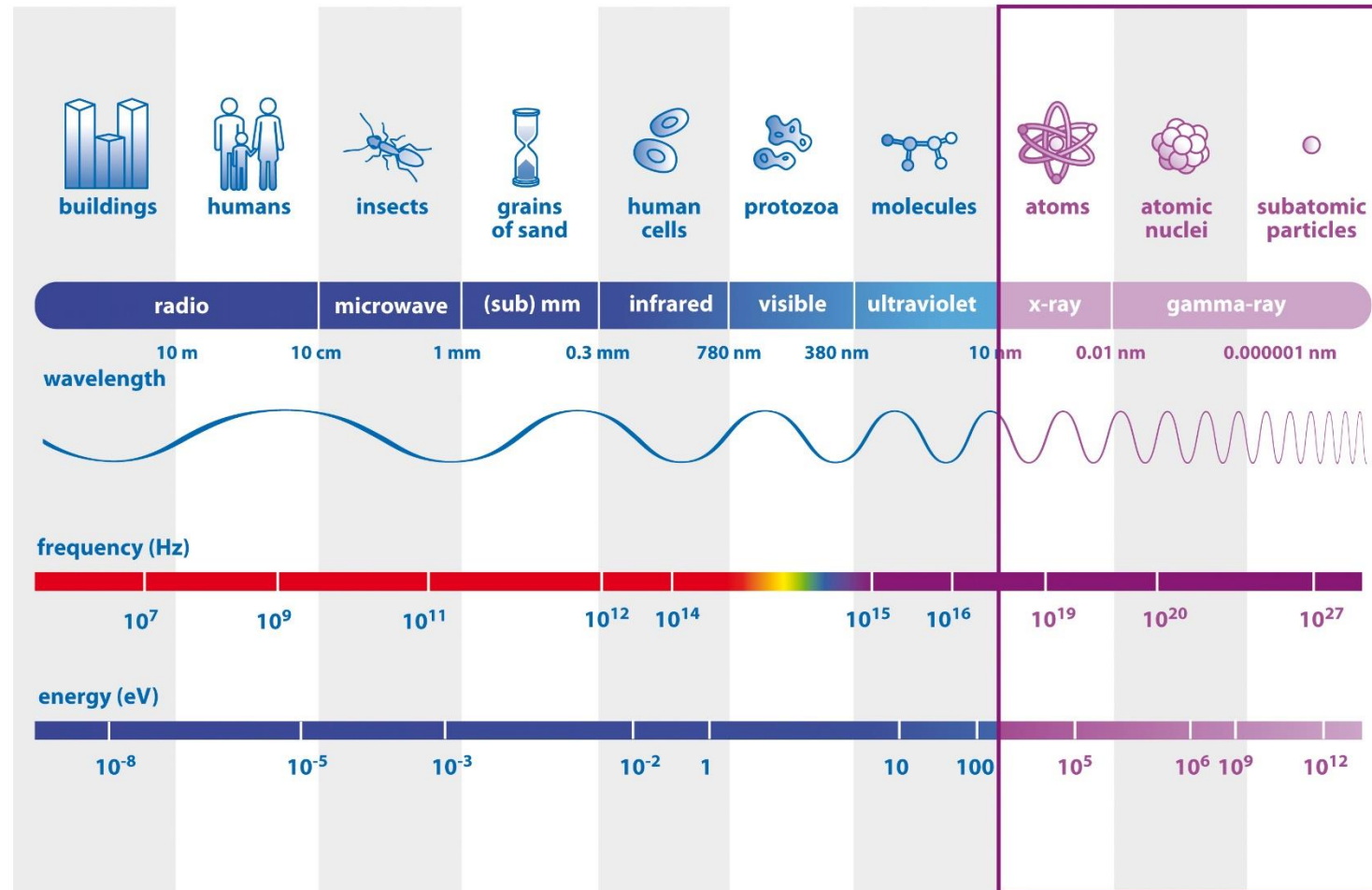
## Literature:

- Vankó György: A szinkrotronsugárzás újszerű kémiai alkalmazásai: nagyfelbontású röntgenspektroszkópia - A kémia újabb eredményei 100. (Budapest, 2008)

# By X-ray spectroscopies here I mean...

- Typically hard X-rays: 4-20 keV  
(but: soft X-rays below 1kV and tender regime between soft and hard)
- Absorption and emission of X-ray photons
- High energy resolution: better than the natural broadening of the signal
- Can be used with high spatial resolution
- Can be used with high time resolution
- Can be used to probe magnetic polarization

# Spectroscopy with (hard) X-rays



NMR ( $\delta$ )

FTMW (MHz)

FTIR ( $\text{cm}^{-1}$ )

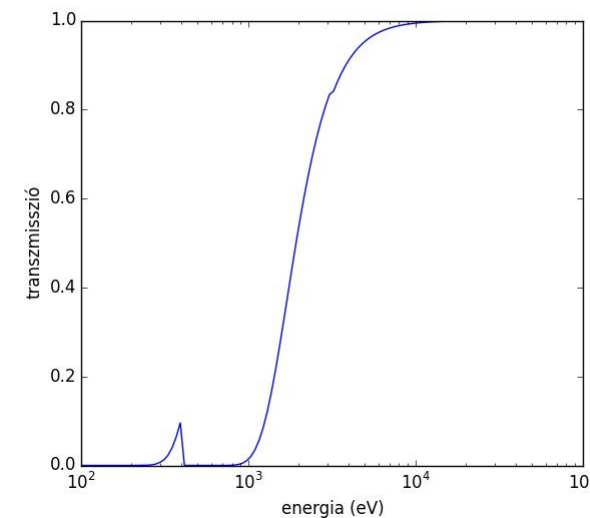
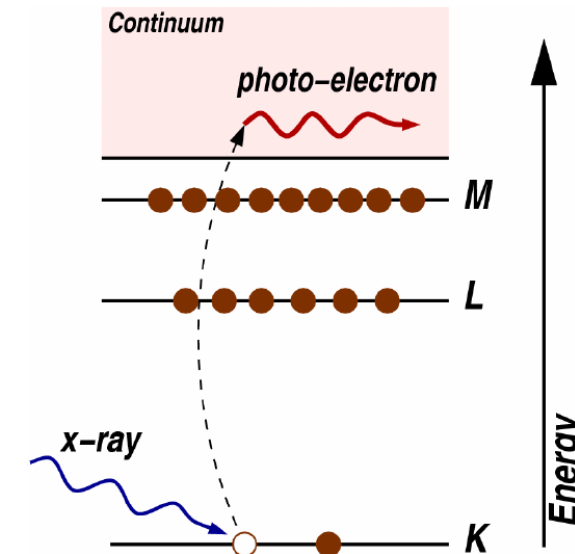
UV-Vis (nm)

X-ray spectroscopies (eV)

Mössbauer spectroscopy (mm/s)

# Hard X-rays

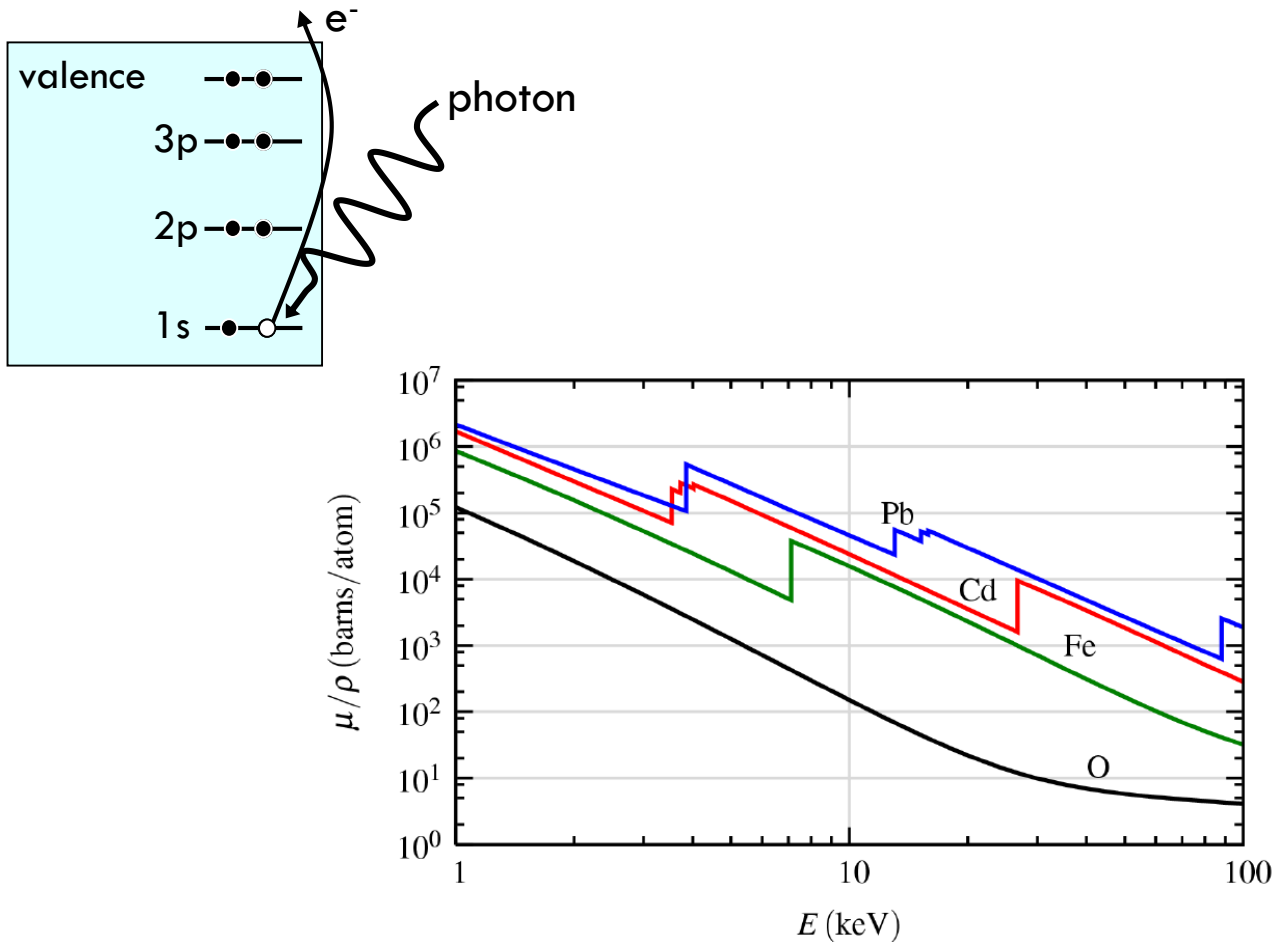
- Typically: photoexcitation of core electrons of 3d/4d metals (e.g. 1s orbital of Fe is 7112 eV, <http://xdb.lbl.gov/>)
- Elemental selectivity: energy resolution is 2 orders of magnitude better than the distance between 2 absorption edges
- High penetration depth (bulk vs. surface)
- Gas phase, liquids, solids
- Not sensitive to air ↔ UPS, XPS, soft X-rays
- But similar methodology for soft or tender X-rays which work for low Z elements like C, N, O or S, Cl, P



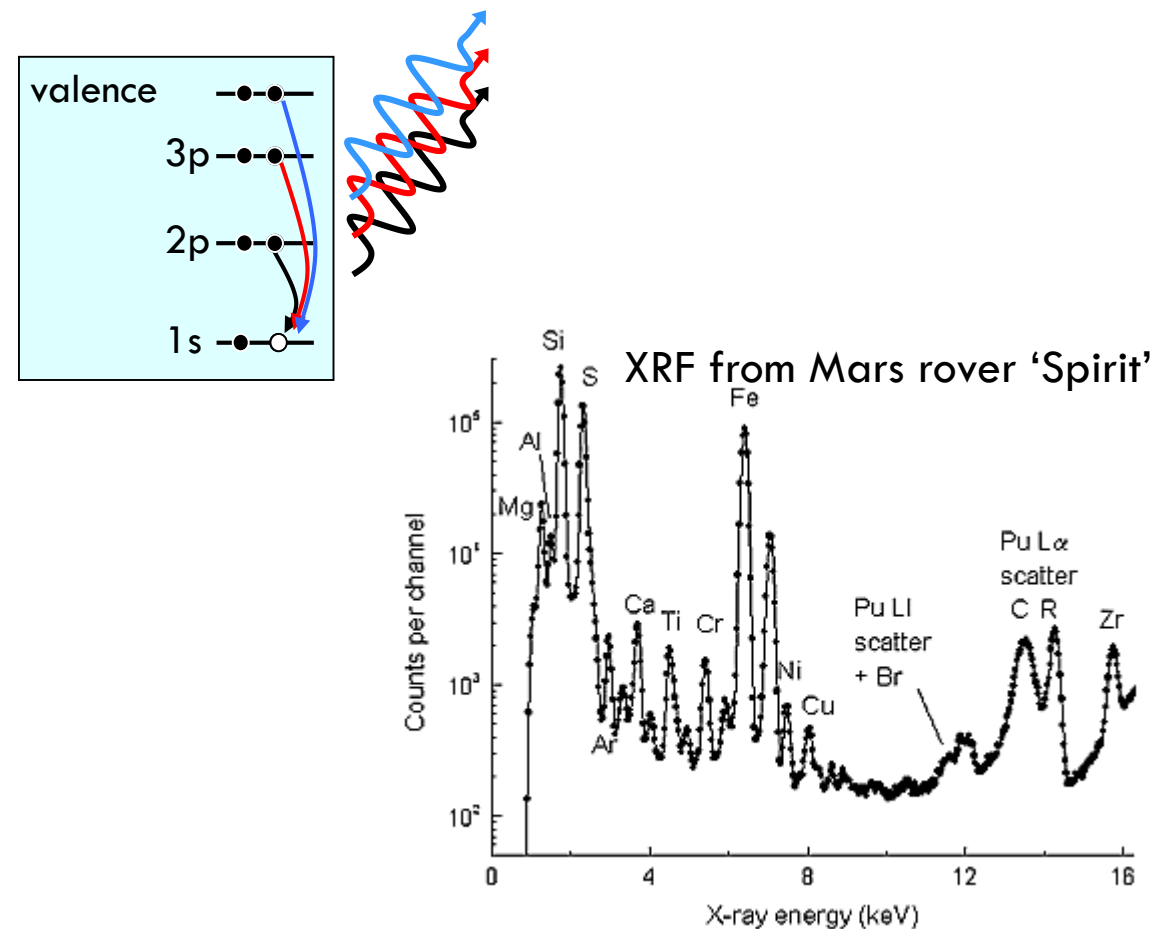
Element	K 1s
23 V	5465
24 Cr	5989
25 Mn	6539
26 Fe	7112
27 Co	7709
28 Ni	8333
29 Cu	8979
30 Zn	9659

# Absorption and emission of X-ray photons

XAS: X-ray absorption spectroscopy



XES: X-ray emission spectroscopy  
(XRF: X-ray fluorescence)



# High energy resolution: reveals the fine structure

- XANES: X-ray absorption near edge structure
- EXAFS: Extended X-ray absorption fine structure

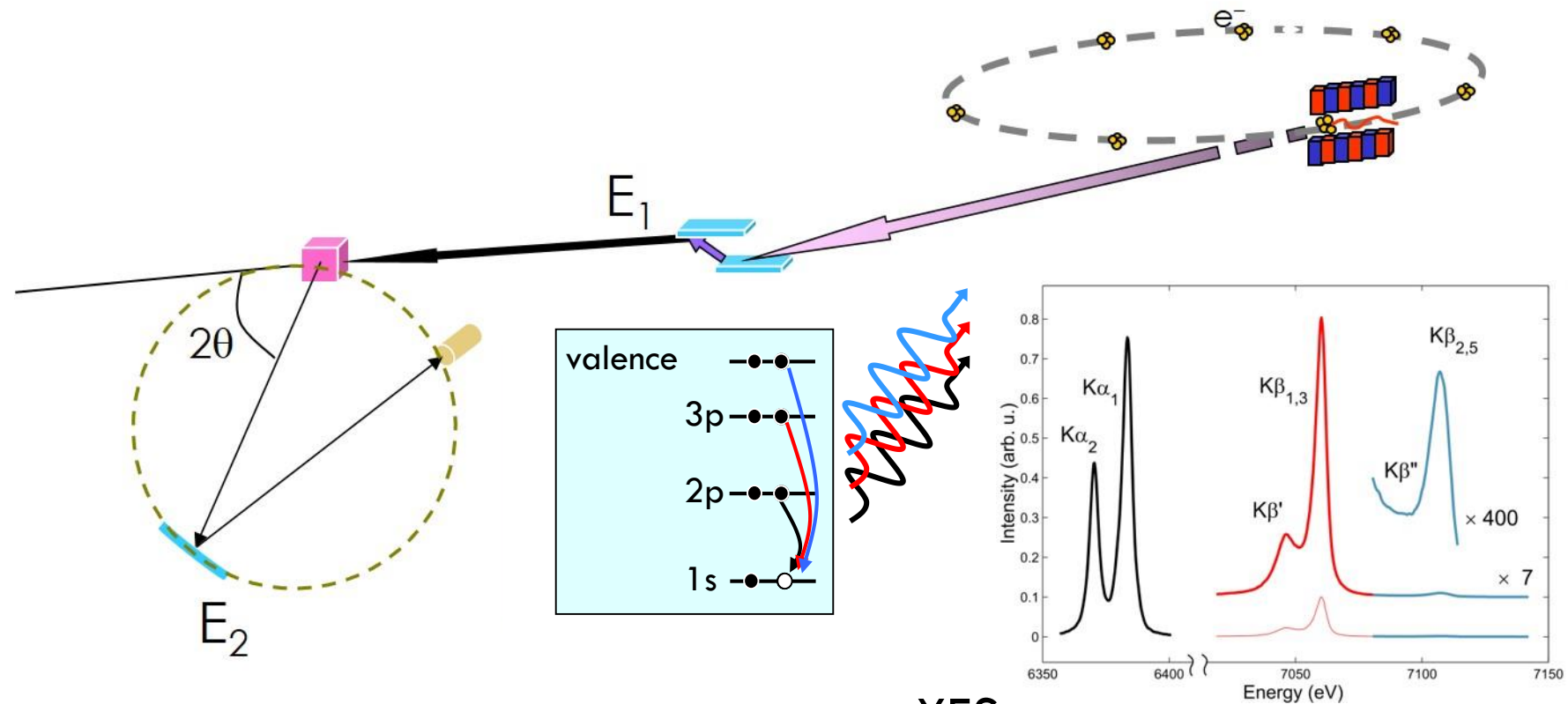
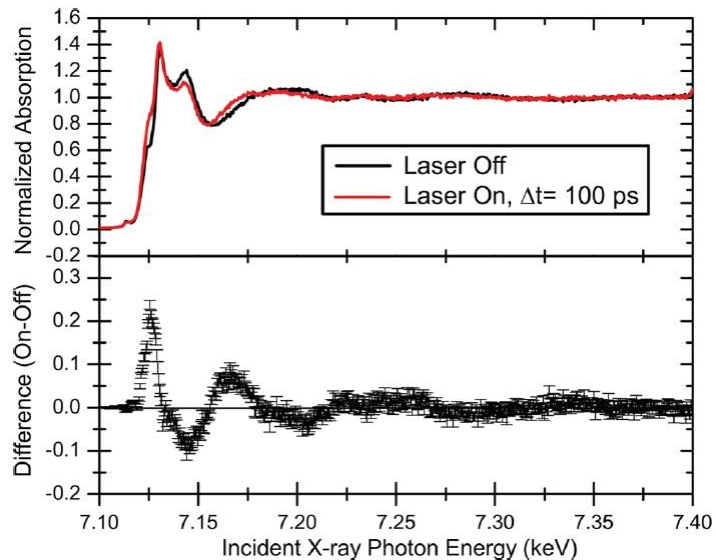
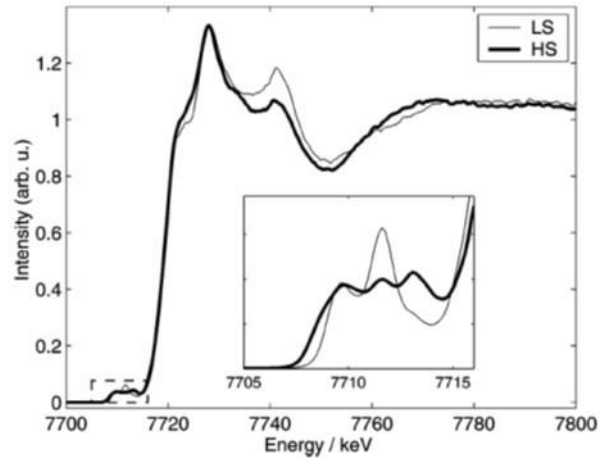


Figure from G. Vankó

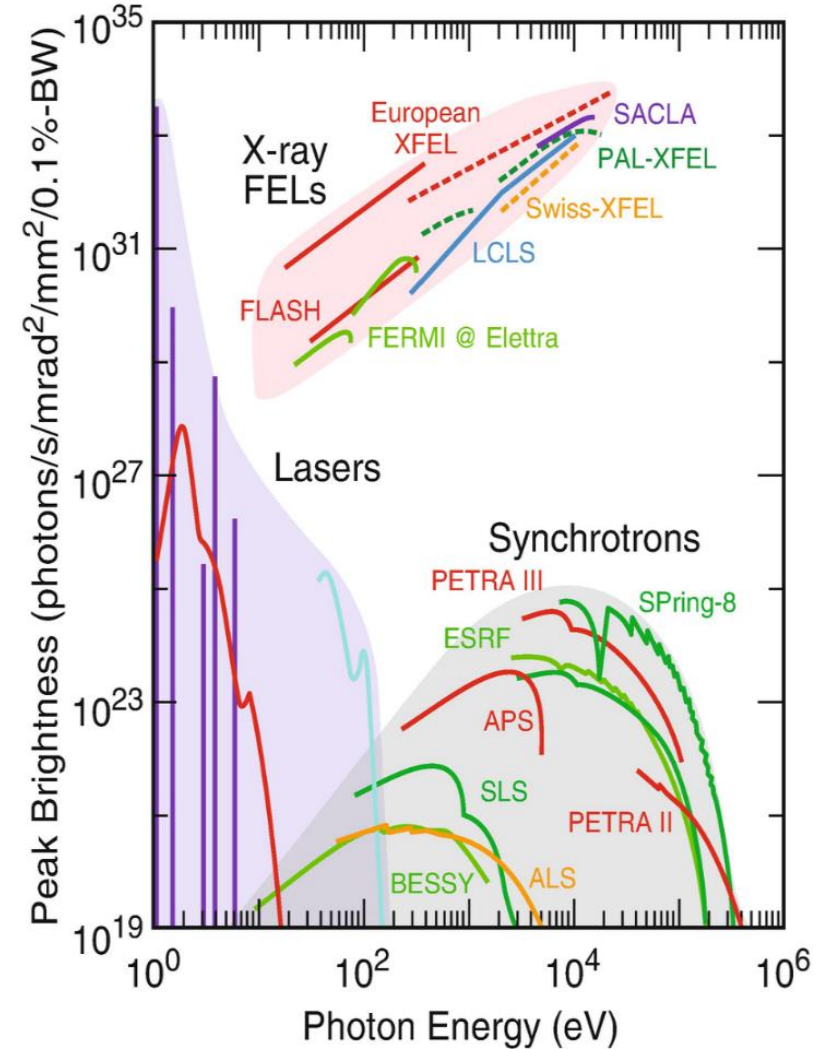
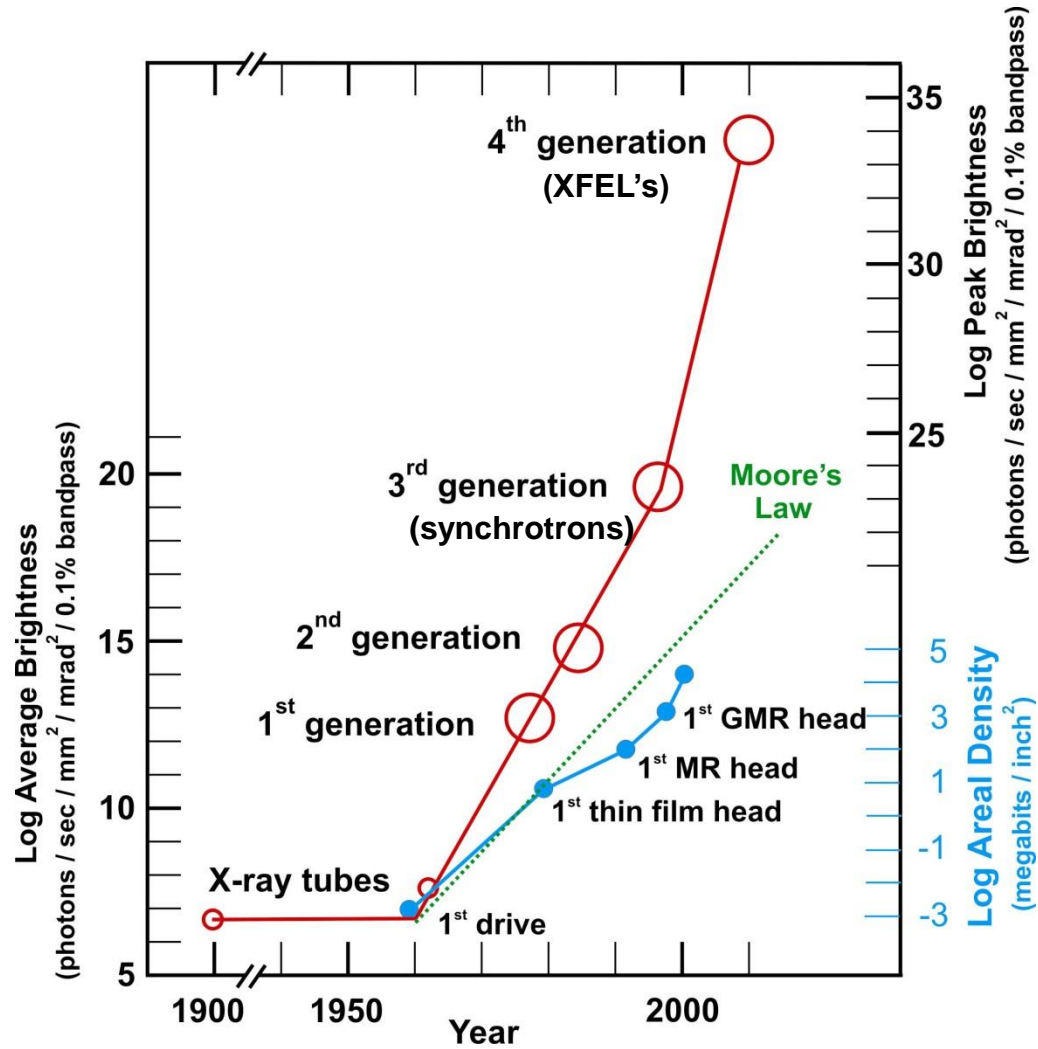
XES



# How to get these spectra?

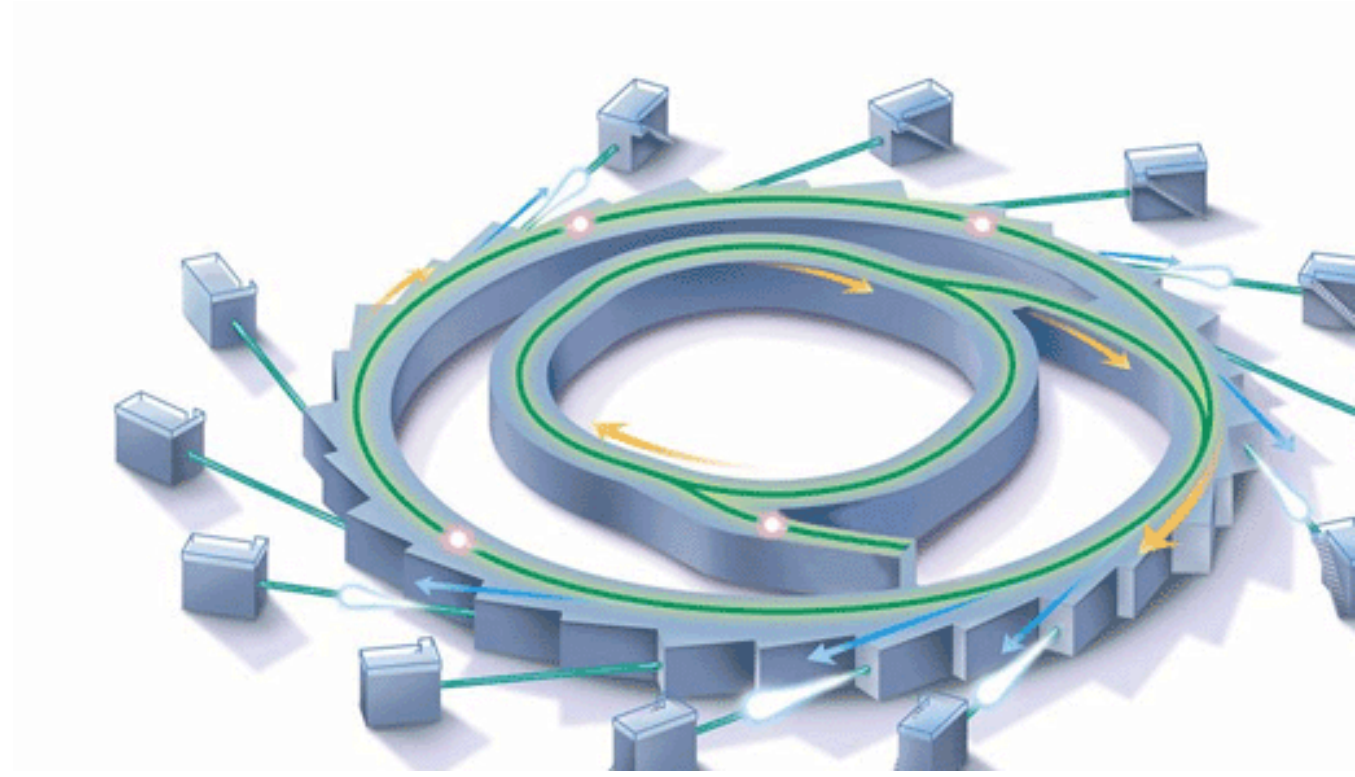
- X-ray sources
  - Synchrotrons
  - XFEL's
  - Laboratory sources
  - Laser based sources
- Detectors
  - Ionization chambers
  - Scintillators
  - Semiconductors
- Resolving by energy (or better wavelength)
  - Internal energy resolution of the detectors
  - Bragg diffraction on single crystals
  - Laue diffraction

# Evolution of X-ray sources



# 3<sup>rd</sup> gen synchrotrons (90's -)

- high brilliance:  $10^6$ – $10^{12}$  times more than conventional laboratory X-ray tubes
- wide energy range (from infrared to hard X-rays)
- usually linearly polarized in the ring plane
- collimated (vertical divergence:  $20 \mu\text{rad}$ )
- pulsed radiation  
(pulse length:  $10^{-11}$ – $10^{-10}$  s,  
period time:  $10^{-6}$ – $10^{-8}$  s (tunable))
- **high demand and high operational cost** ca. €100M/a  
→ €10k/day/experiment



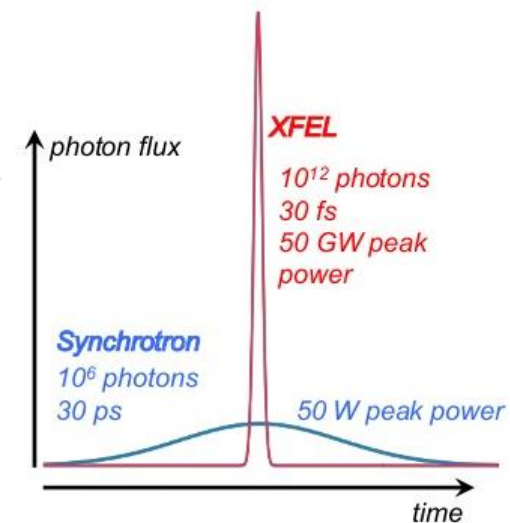
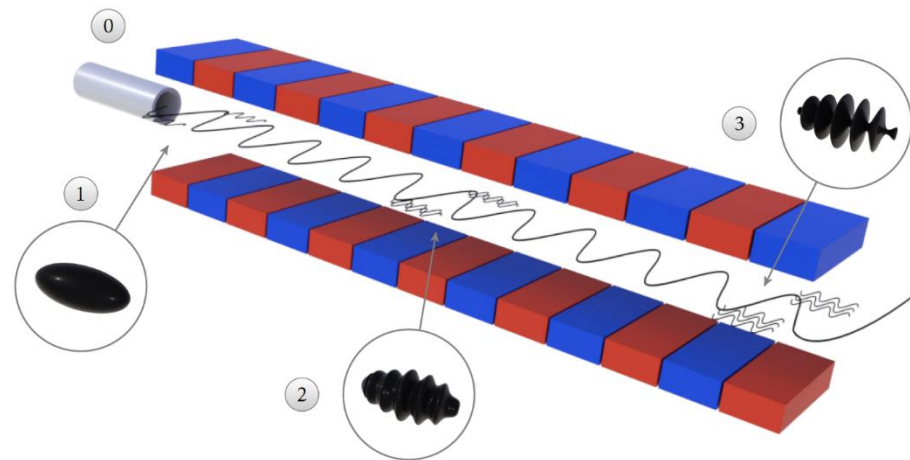
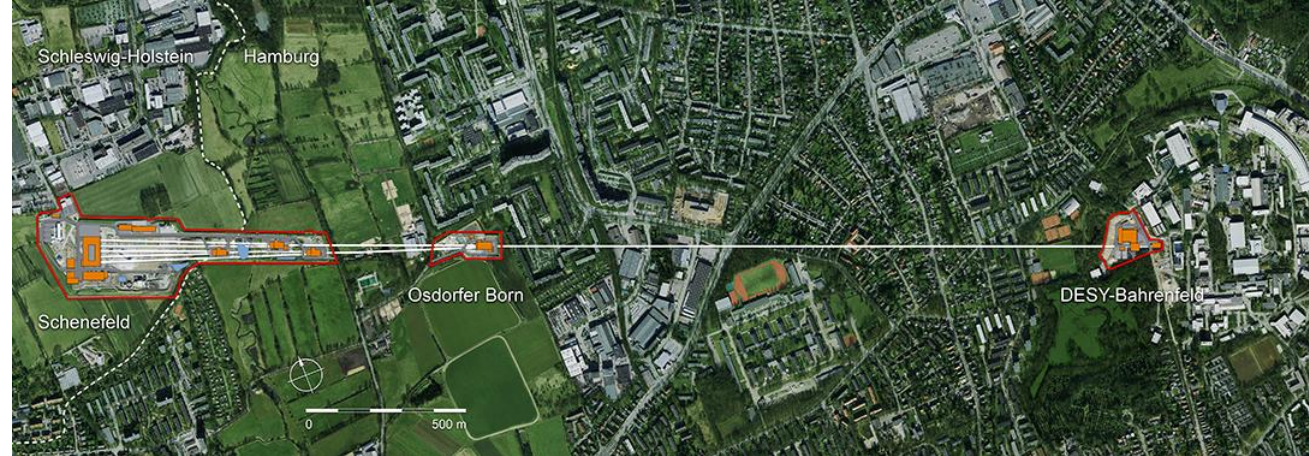
# 3<sup>rd</sup> gen synchrotrons (90's -)

- 4 major synchrotrons (ESRF/EU, PETRA IV/DE, Spring8/Japan, APS/USA) and many smaller ones
- Beamtimes via accepted proposals (2/3 rounds per year, evaluated by international committees)
- Short, condensed beamtimes (few days)
- Local scientific and technical support
- Financial support through EU agencies for EU beamlines
- <https://lightsources.org/>



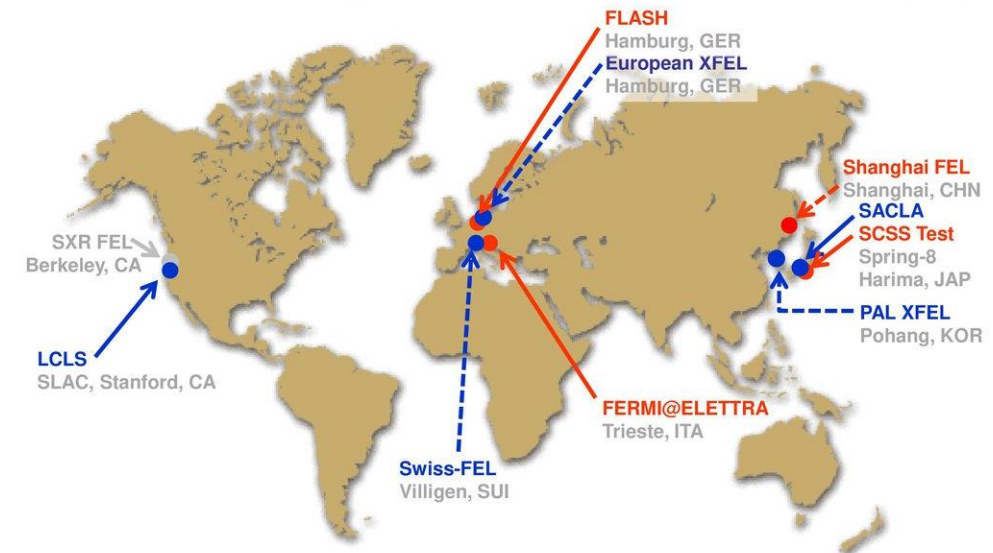
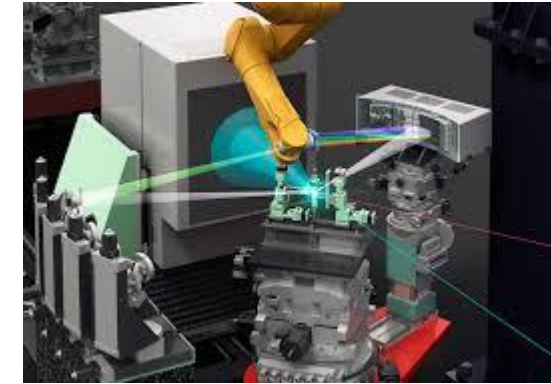
# X-ray free electron lasers (10's -)

- Long linear accelerators (few km), sometimes made of superconductors
- Nearly the speed of light electrons are driven through special magnets (undulators) to emit photons
- SASE (self-amplified spontaneous emission): the coherent electron and photon bunches align each other to form ultrashort pulses
- Result:  $\sim 10$  fs long extreme intense ( $10^8$  more peak brightness than synchrotrons) hard X-ray bunches



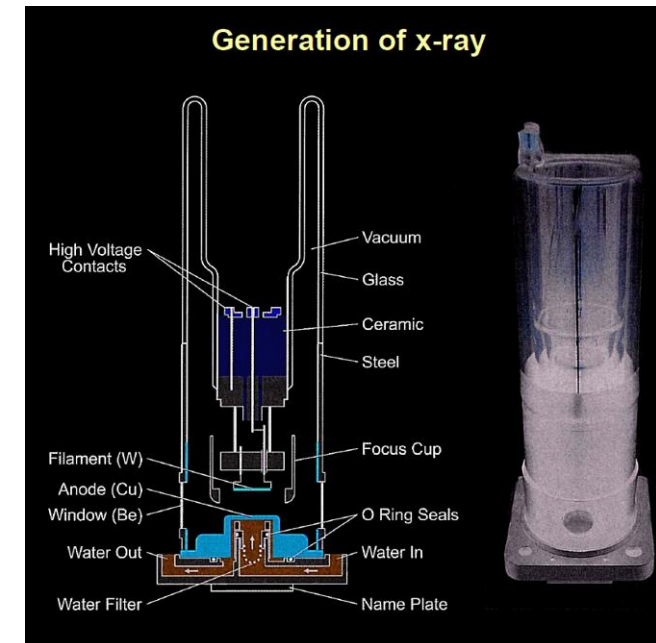
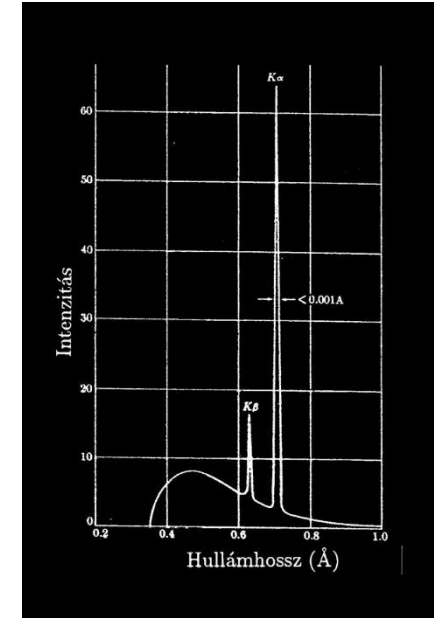
# X-ray free electron lasers (10's -)

- 5 hard X-ray free electron lasers in operation (EuXFEL/EU, SwissFEL/CH, SACLA/Japan, PAL-XFEL/Korea, LCLS/USA)
- Extreme high load on the beamlines
- **€1B construction cost**
- **€1M/day/experiment**

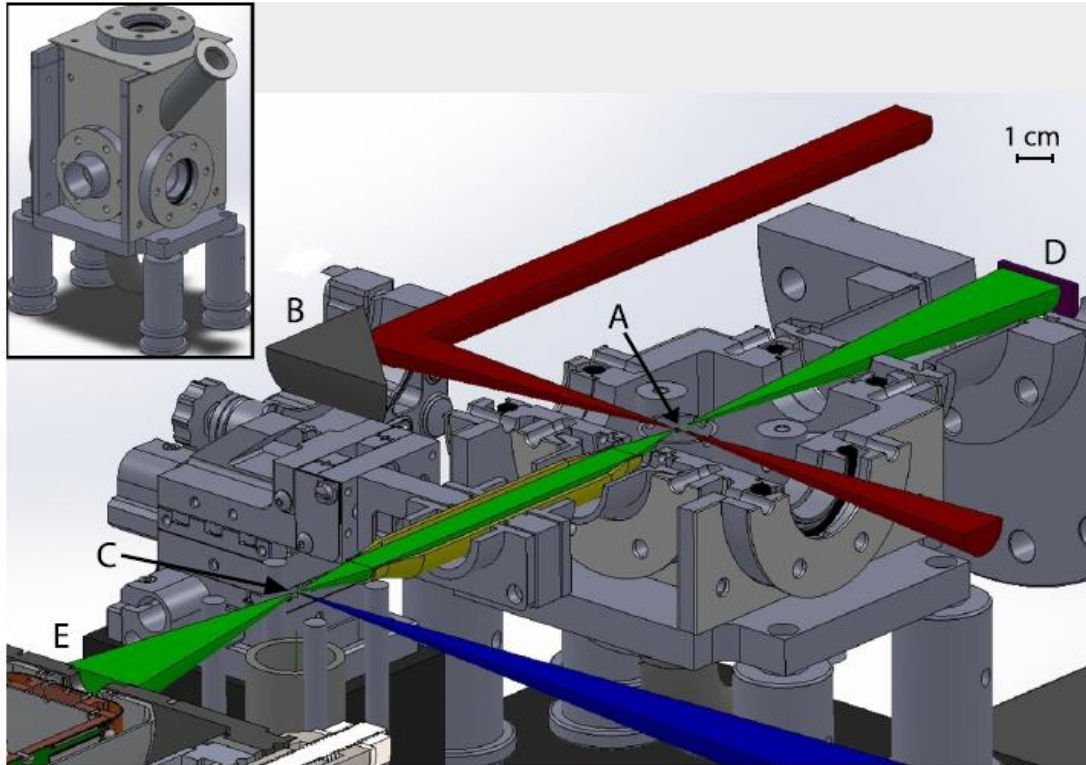


# Laboratory X-ray sources

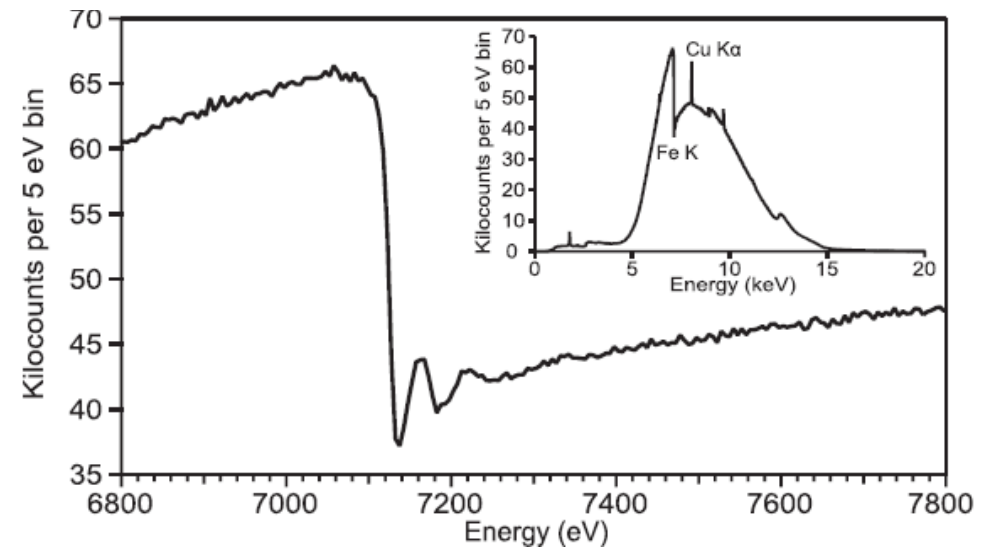
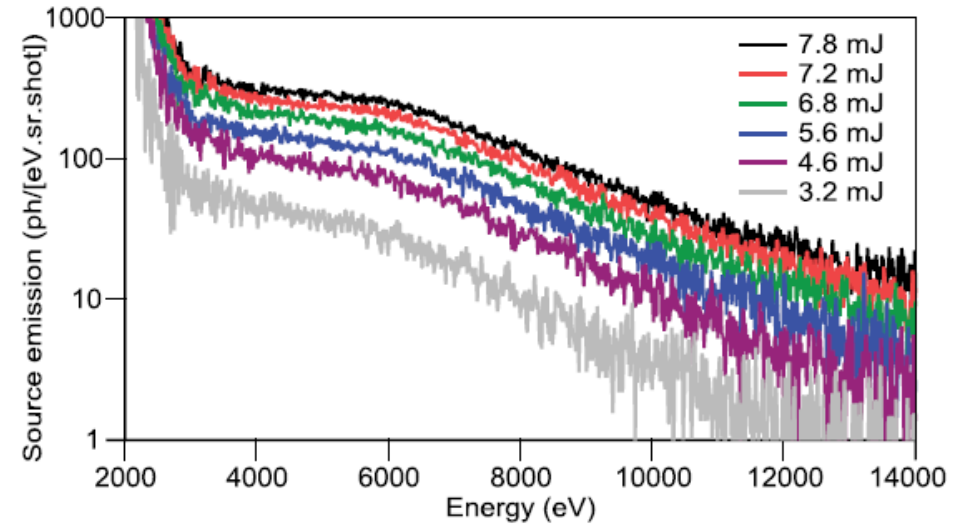
- **€0.01M investment, negligible running costs**
- Stable, well controlled intensity for long runs
- Can be focused to  $<100\mu\text{m}$
- Not polarized, no time structure (CW)
- Broad bremsstrahlung + characteristic peaks of the anode
- Intended use: diffraction, imaging, XRF



# Laser plasma sources



- A: water jet
- B: mirror focusing the incident laser beam
- C: sample
- D: beam monitor
- E: detector



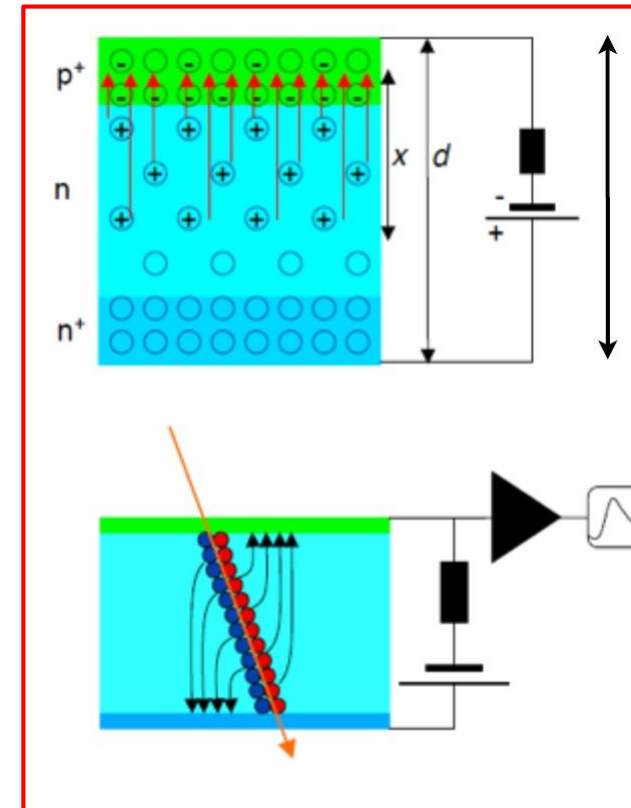
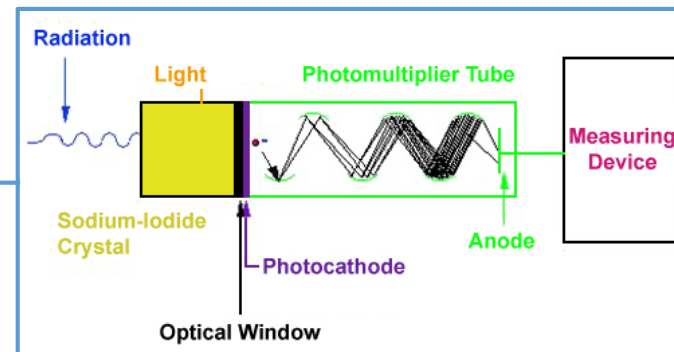
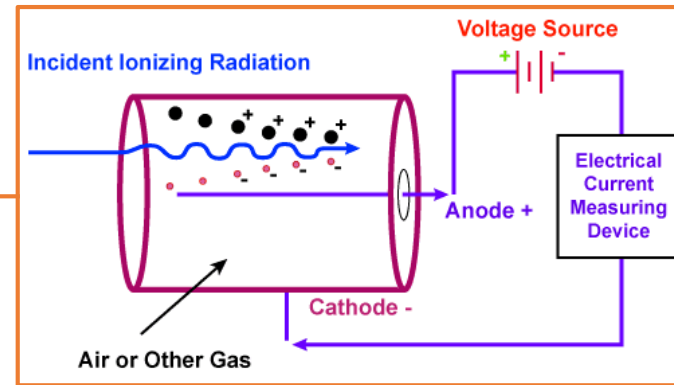
W. Fullagar, et al, *Rev. Sci. Instrum.* 78, 115105 (2007), Lund, Sweden

L. Miaja-Avila, et al., *Struct. Dyn.* 2, 024301 (2015), Boulder, Colorado, USA



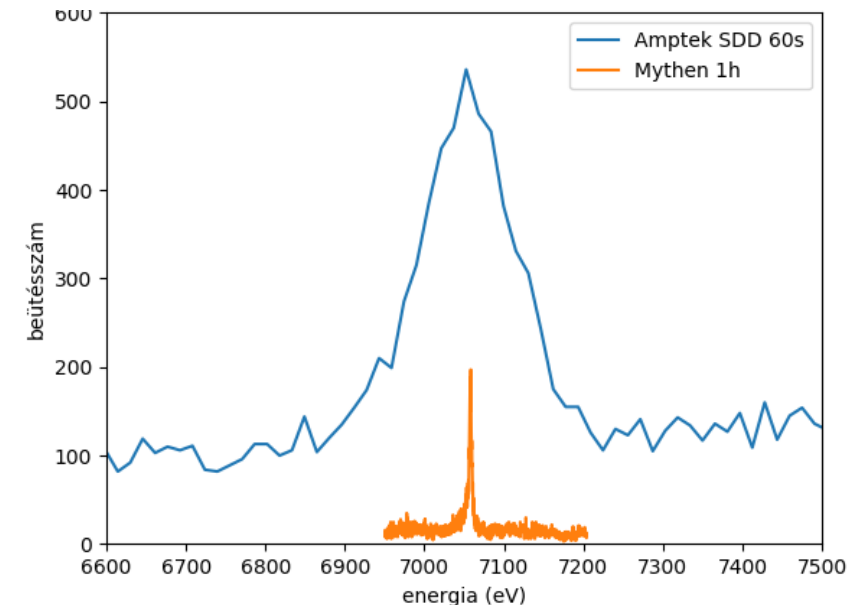
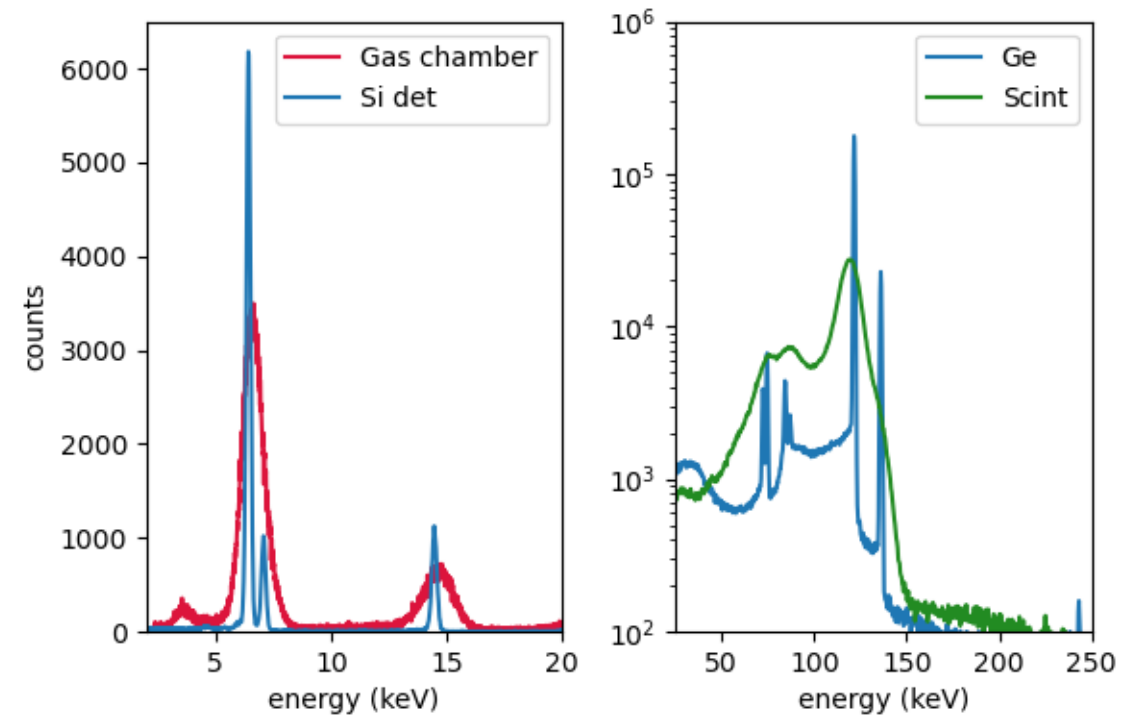
# Detection

- Ionization chambers:
  - Handle high photon fluxes
  - Good proportionality
- Scintillation detectors
  - Moderate energy resolution
  - Good efficiency
  - Slow
- Semiconductors
  - High energy resolution
  - Good efficiency
  - Fast (low deadtime)
  - Low background
  - Scalable – can easily be used for building 2D arrays



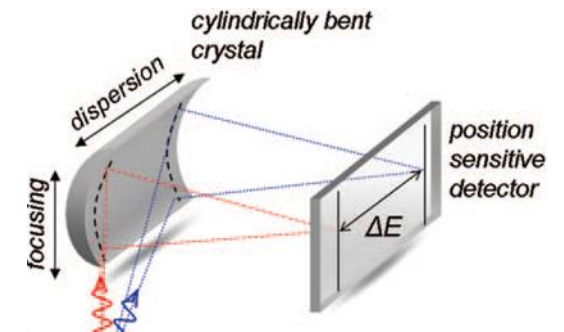
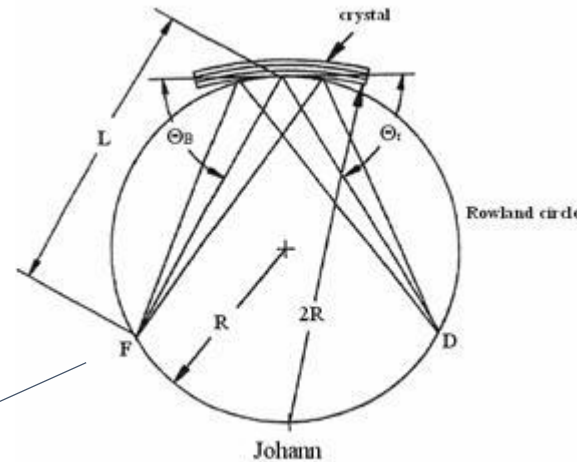
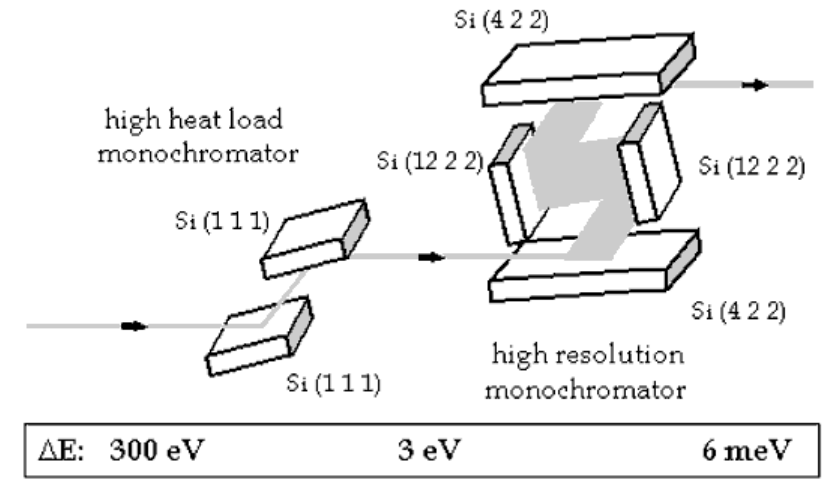
# Resolving by energy

- Internal energy resolution of the detectors: discriminating by the signal height of the amplified electronic signal
- Depends strongly on the detector's working principle
- Typically: scintillators < ion chambers < semiconductors
- Best energy resolution: Silicon Drift Diodes (SDD), ca.  $10^2$  eV at several kV's ( $= 10^{-1} E/E_0$ )
- But! Fine structure needs at least  $10^{-3} E/E_0$



# Resolving by wavelength

- Bragg diffraction or Laue diffraction
- Well know ( $d$ ) diffracting single crystal analyzer
- Photons with different wavelengths will diffract at different Bragg angles
- Subsequent monochromators at high intensity synchrotron beamlines
- Smaller single crystals (Si, Ge, etc.) in spectrometers
  - Scanning mode (Rowland circle, Johann geometry with spherically bent crystal)
  - Static mode (von Hámos geometry with cylindrical crystal)



# 2. X-ray Absorption Spectroscopies

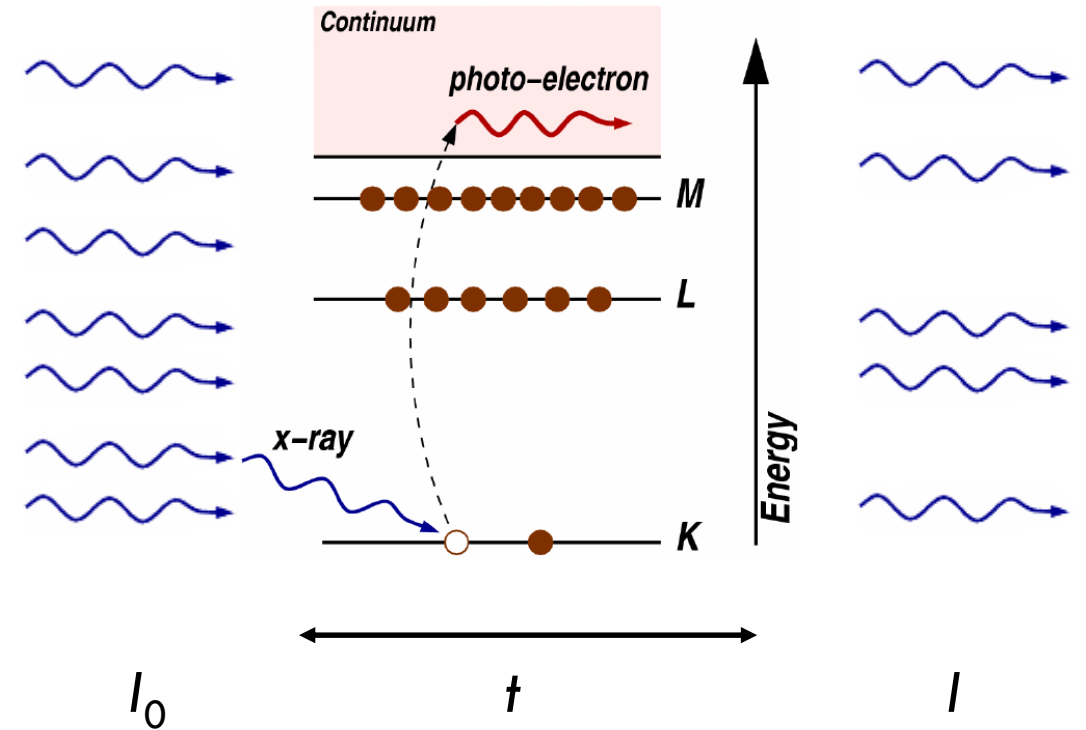
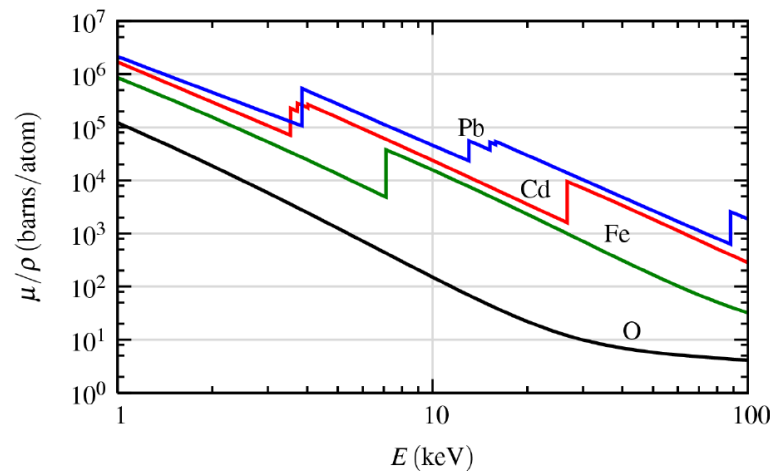
## Literature:

- M. Newville: Fundamentals of XAFS ,Reviews in Mineralogy & Geochemistry Vol. 78 pp. 33-74, 2014
- <http://gbxafs.iit.edu/training/xafsoverview.pdf>
- *X-Ray Absorption – Principles, Applications, Techniques of EXAFS, SEXAFS and XANES*, ed. by D. C. Koningsberger and R. Prins, John Willey (1988)
- A. L. Ankudinov, B. Ravel, J. J. Rehr, S. D. Conradson, Phys. Rev. B 58, 7565 (1998)
- F. de Groot: High-Resolution X-ray Emission and X-ray Absorption Spectroscopy, *Chem. Rev.* 2001, 101, 1779-1808

# The X-ray absorption effect

- Photo-electric effect (Albert Einstein, Nobel Prize in Physics, 1921)
- Beer–Lambert law:  $I = I_0 e^{-\mu t}$ ,
- Quantity to measure:  $\mu$  vs.  $E$

- $\mu \approx \frac{\rho Z^4}{AE^3}$



# The X-ray absorption coefficient

- Photoabsorption via Fermi's golden rule:

$$\mu \propto |\langle f | \hat{e} \cdot r e^{ik \cdot r} | i \rangle|^2 \delta_{E_f - E_i - \hbar\omega}$$

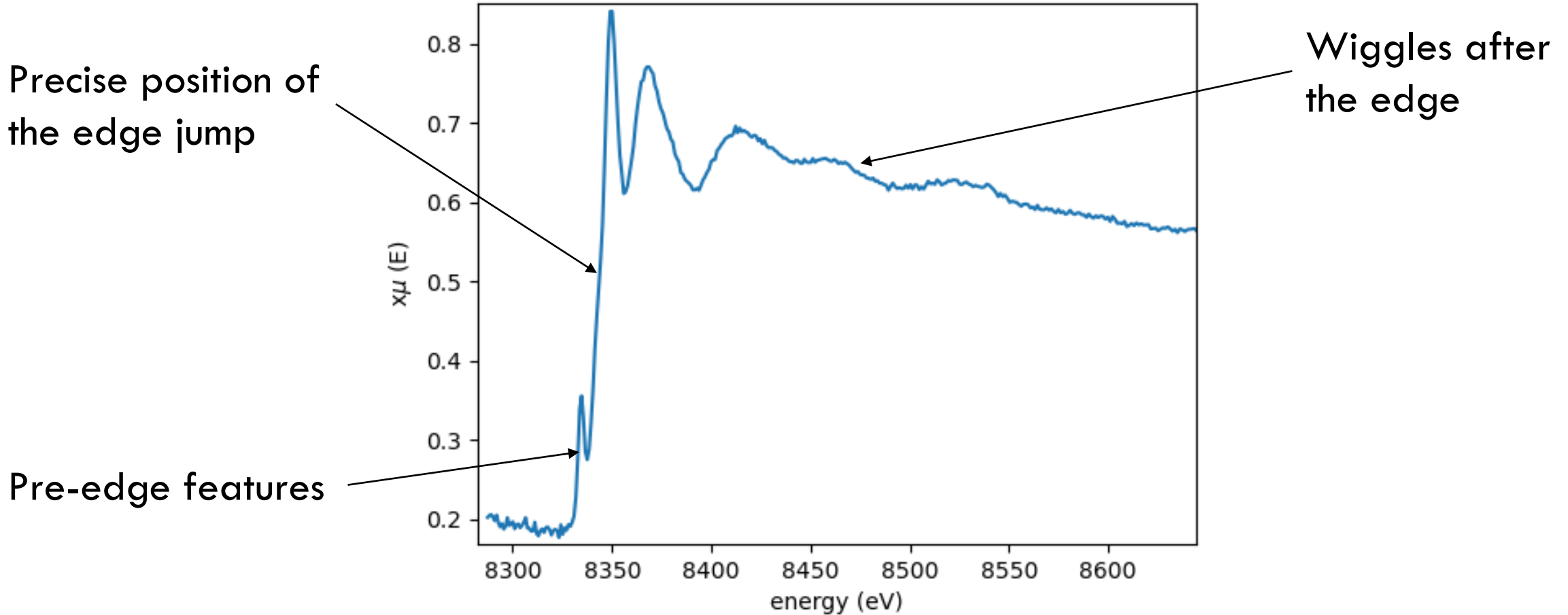
- Final state = initial state with a continuum electron ( $\epsilon$ ) added and a core electron removed ( $c$ )
- All inactive electrons neglected, thus the series of delta functions become the density of states ( $\rho$ ):

$$\mu \propto |\epsilon | \hat{e} \cdot r | c \rangle|^2 \cdot \rho$$

- XAS intensity becomes a measure of the DOS

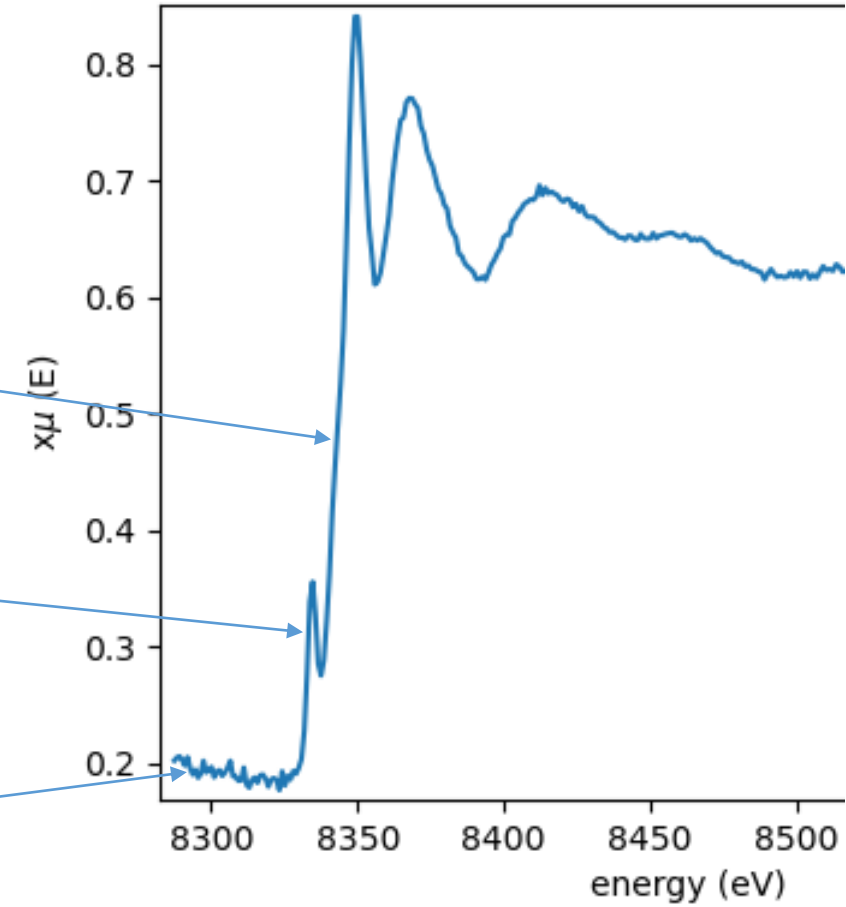
# What can be seen with high energy resolution?

Ni K edge XAS of  $[\text{Ni}(\text{CN})_4]^{2-}$



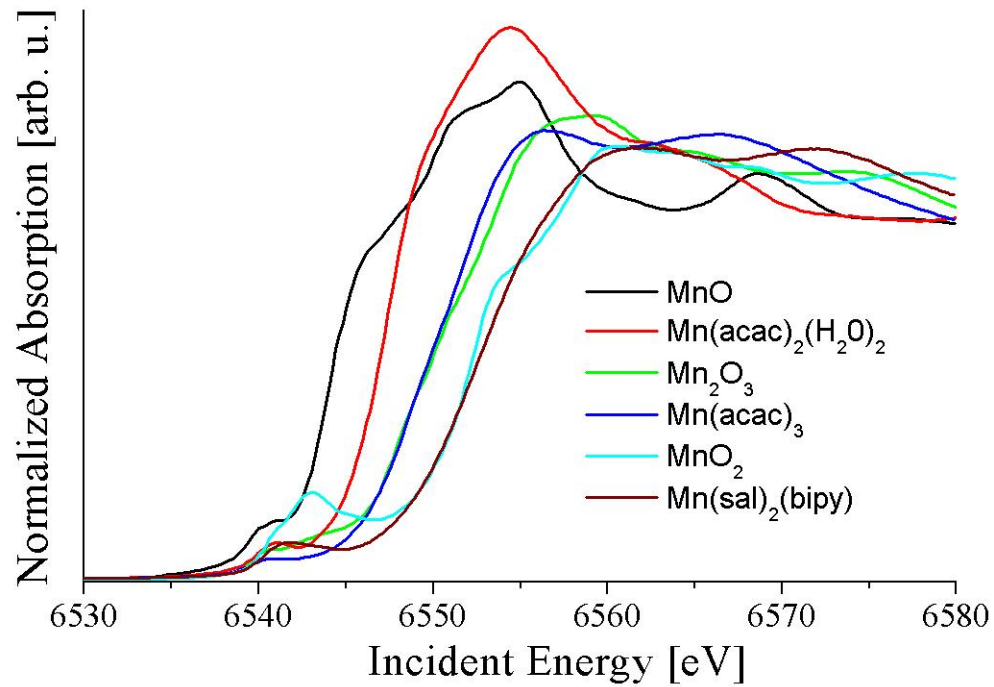
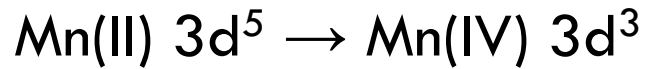
# XANES – region near the absorption edge

- Dipole transition matrix is ca. 137 times stronger than quadrupole, typical case is to excite 1s electron to empty 4p orbital in 3d transition metals (or 2p → 5d)
  - Edge position reflects the lowest lying empty orbitals (LUMO or conduction band)
  - Pre-edge peaks appear due to mixing the quadrupole and dipole orbitals (1s → 3d'ish)
- Absorption is small if  $\hbar\omega < (E_f - E_i)$ 
  - Baseline before the edge

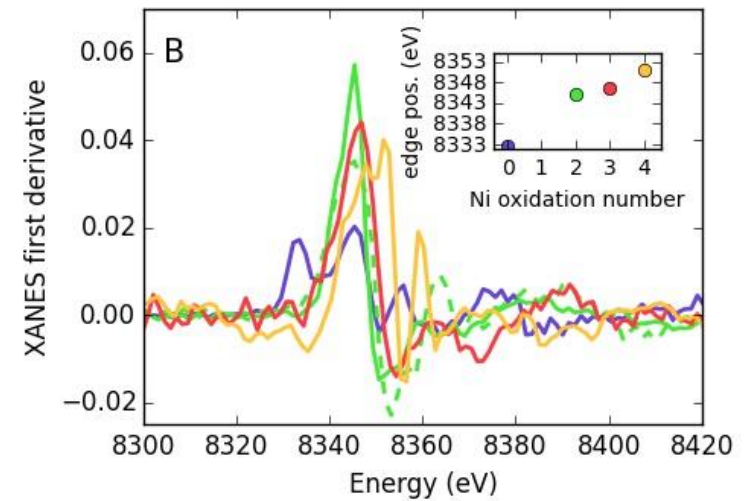
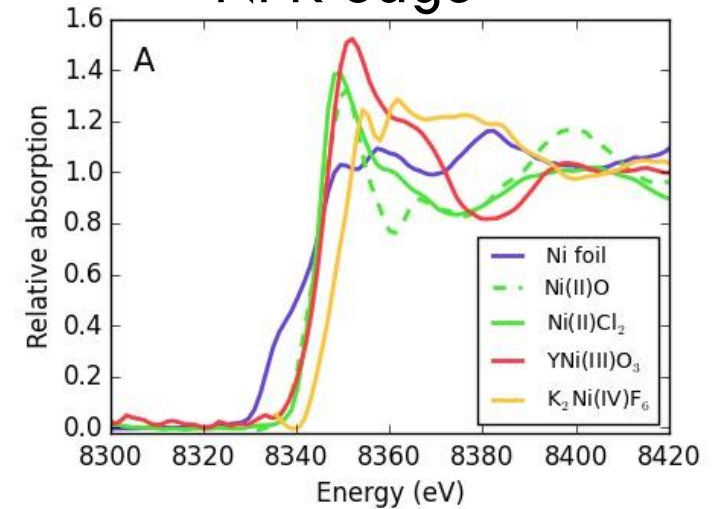




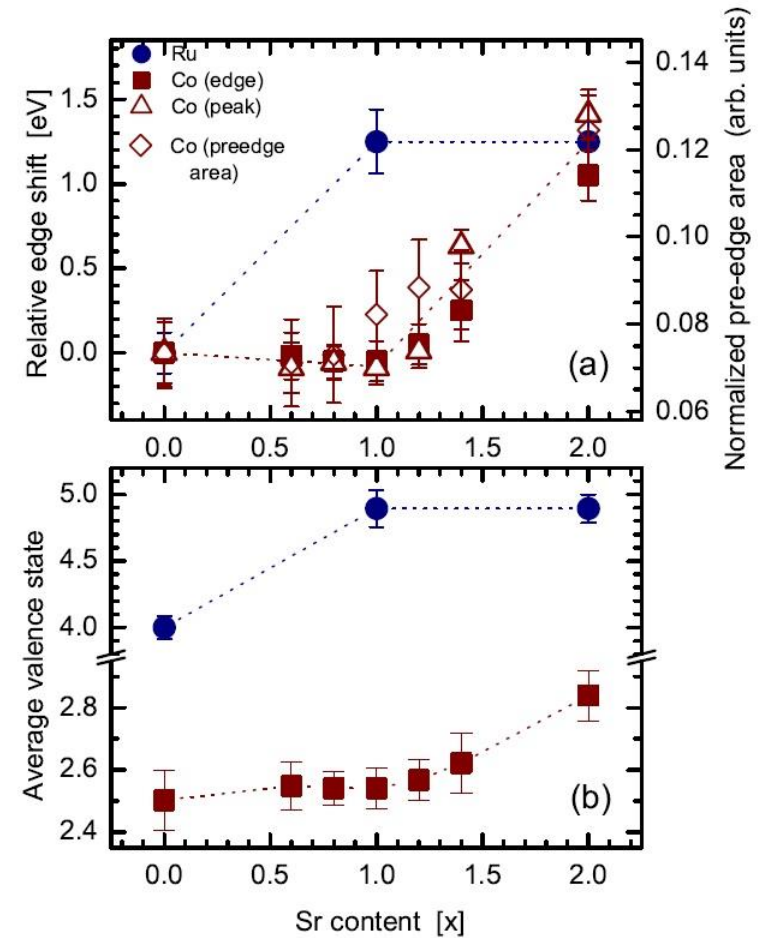
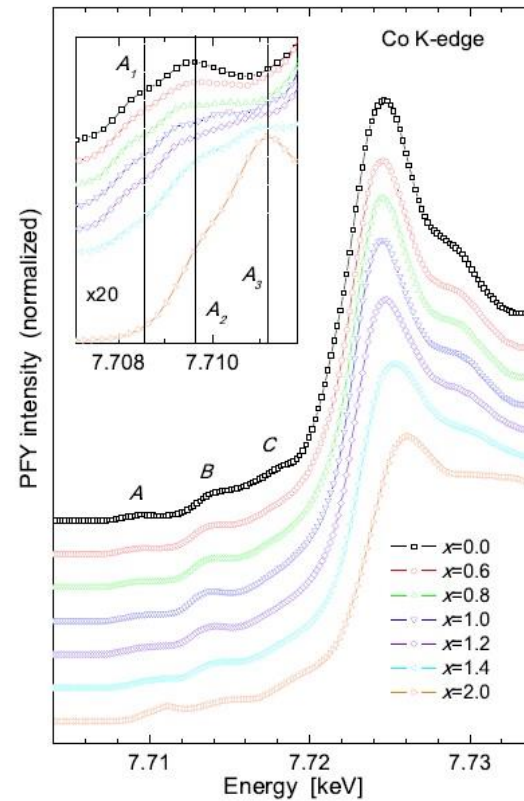
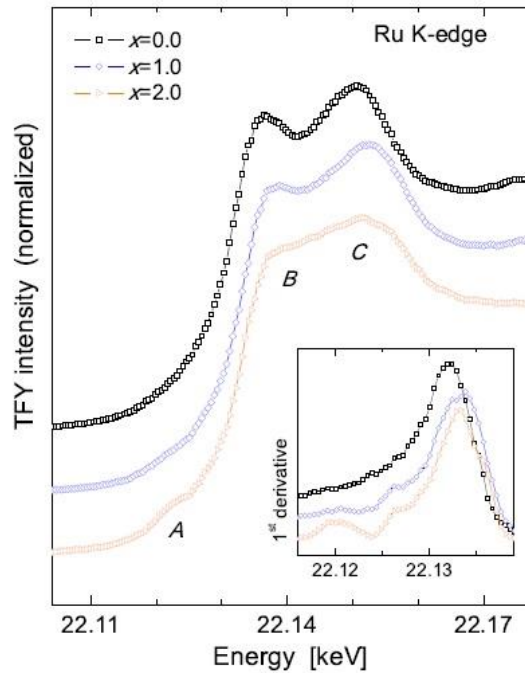
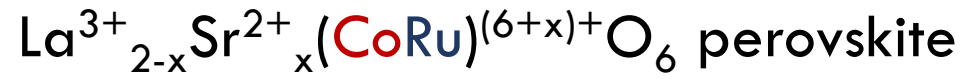
# XANES – typical applications



## Ni K edge

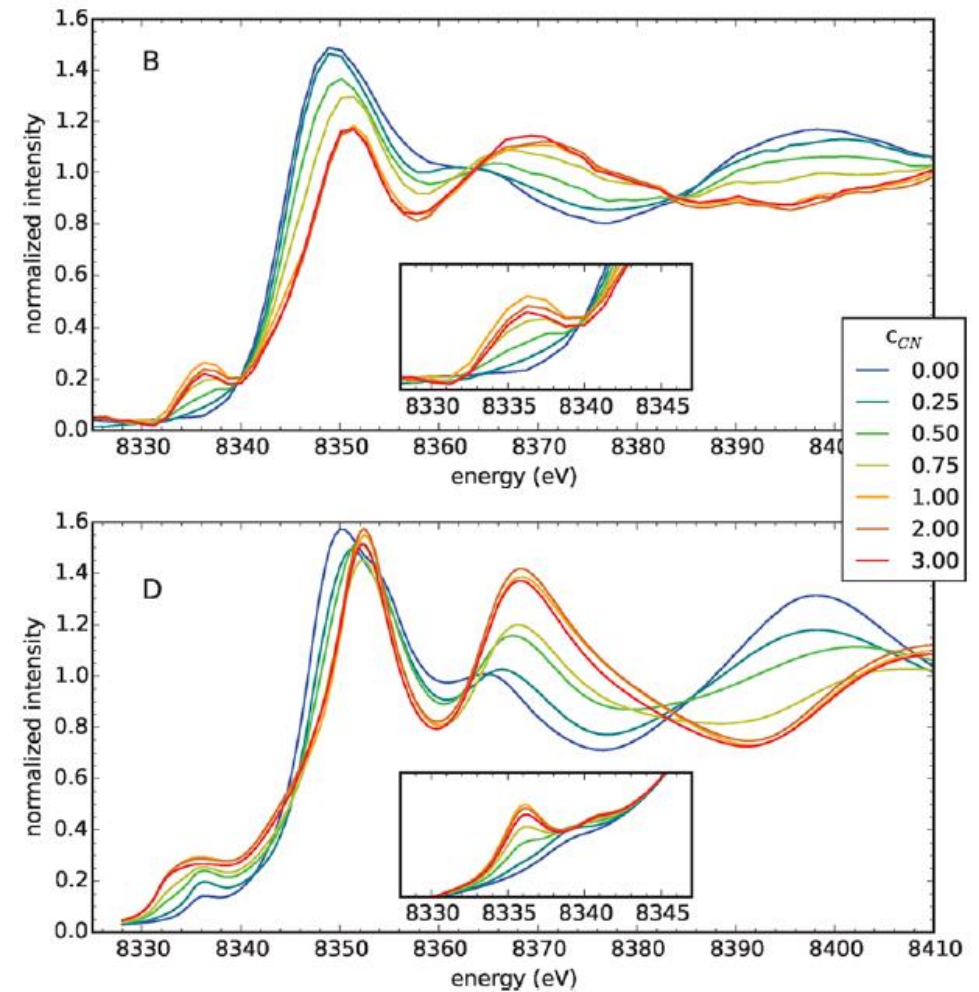
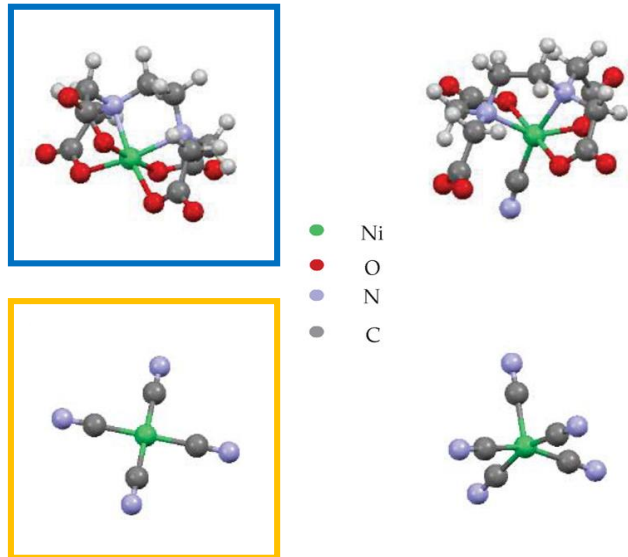


# XANES – typical applications



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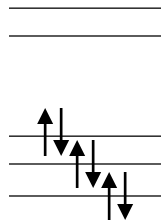
- Pre-edge vanishing with increasing symmetry
- From **square planar** to **octahedral** Ni<sup>2+</sup>



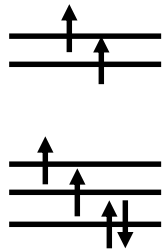
# XANES – spin state dependence

- Not only the density of states, but also the spin density affects the fine structure of the XANES spectrum
- An example:  $\text{Co}^{3+}$  ion ( $3d^6$ ) in

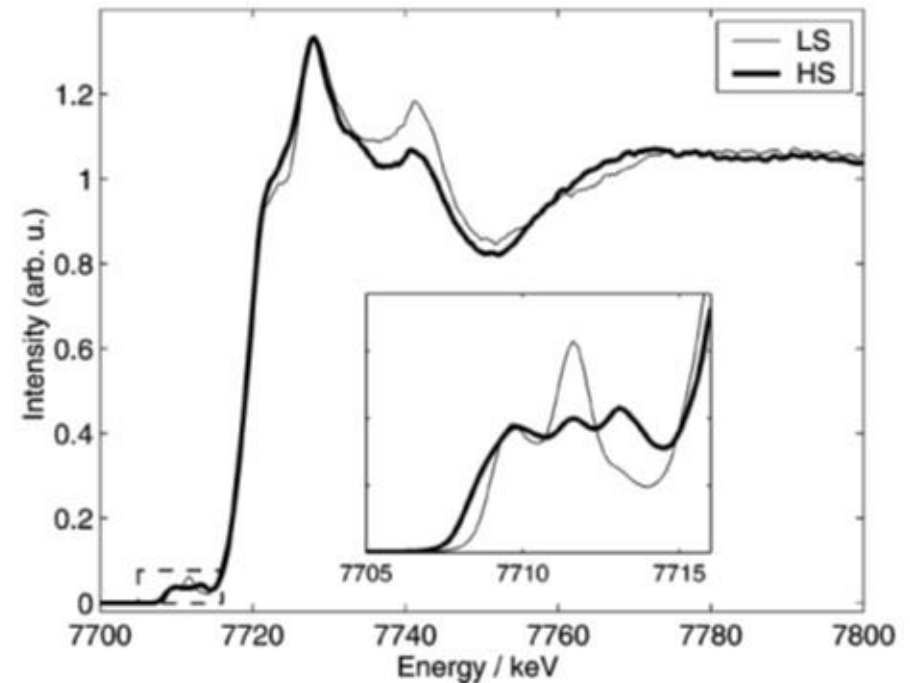
low spin state ( $t_{2g}^6 e_g^0$ )



and high spin state ( $t_{2g}^4 e_g^2$ )

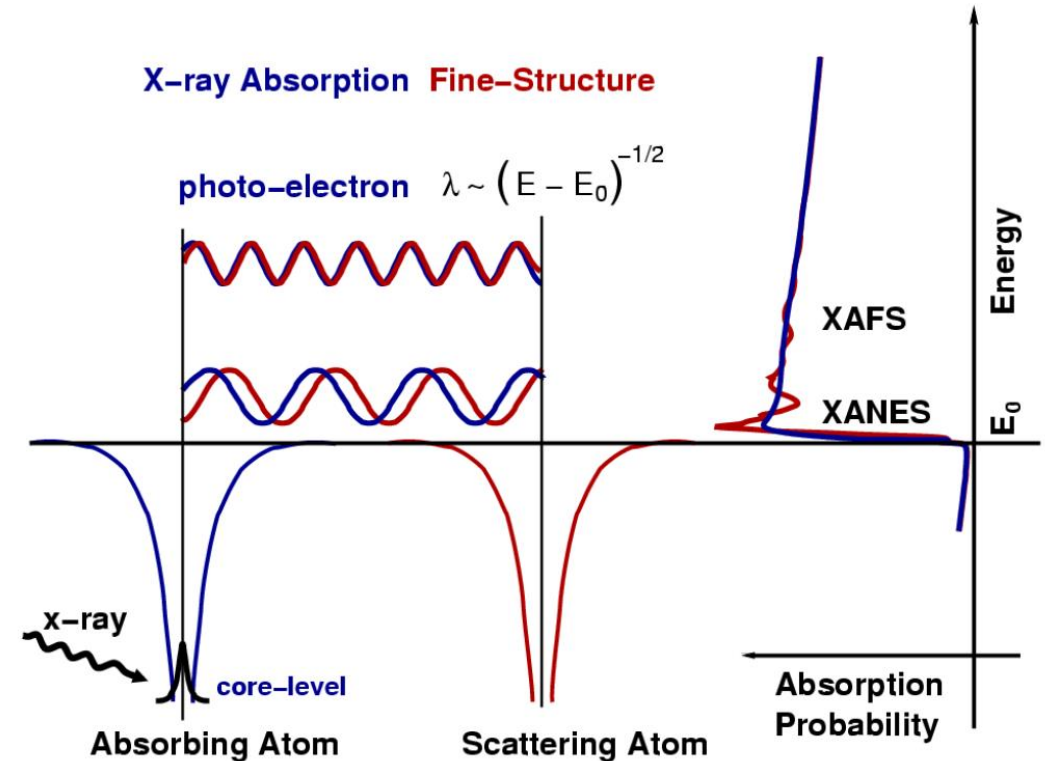


in  $\text{LaCoO}_3$  perovskite



# EXAFS – behind the absorption edge

- If the incident X-ray energy is high enough, the photoelectron will be excited to the continuum with  $\lambda$
- If the excited atom has no neighbors, the absorption coefficient is not perturbed (blue line)
- The photoelectron wave can scatter back from the neighboring atoms and modify the absorption coefficient resulting in an interference pattern (red line)



# The EXAFS equation

- Removing the isolated atom contribution ( $\mu_0$ ) and the edge jump gives the fine structure function:

$$\chi(E) = \frac{\mu(E) - \mu_0(E)}{\Delta\mu_0(E)}$$

- It is converted to wave number scale  $\chi(k)$  and usually multiplies by  $k^2$  or  $k^3$  to account for the dumping nature of  $\chi(k)$

- The final equation is:

$$\chi(k) = \sum_i N_i F_i(k) \frac{S_0^2}{kR_i^2} e^{\frac{-2R_i}{\lambda}} e^{-2\sigma_i^2 k^2} \sin(2kR_i + \varphi_i(k))$$

Scattering amplitude

Damping

Disorder

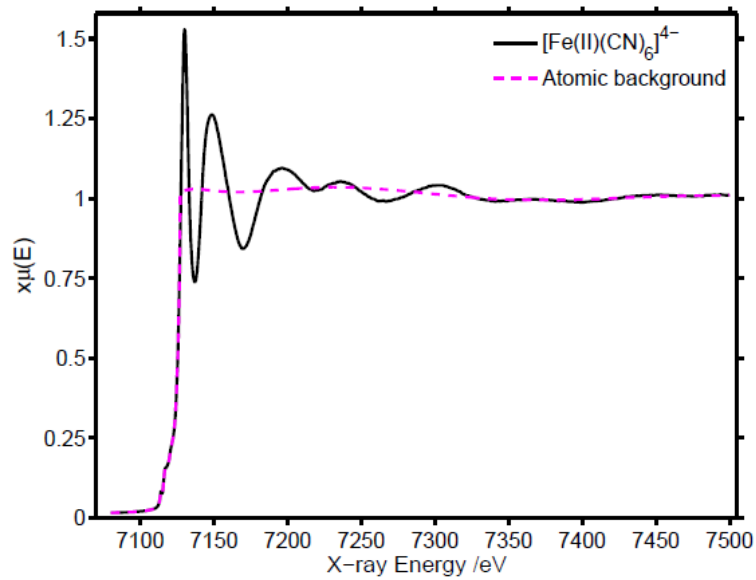
# The EXAFS equation

$$\chi(k) = \sum_i N_i F_i(k) \frac{S_0^2}{k R_i^2} e^{-\frac{2R_i}{\lambda}} e^{-2\sigma_i^2 k^2} \sin(2kR_i + \varphi_i(k))$$

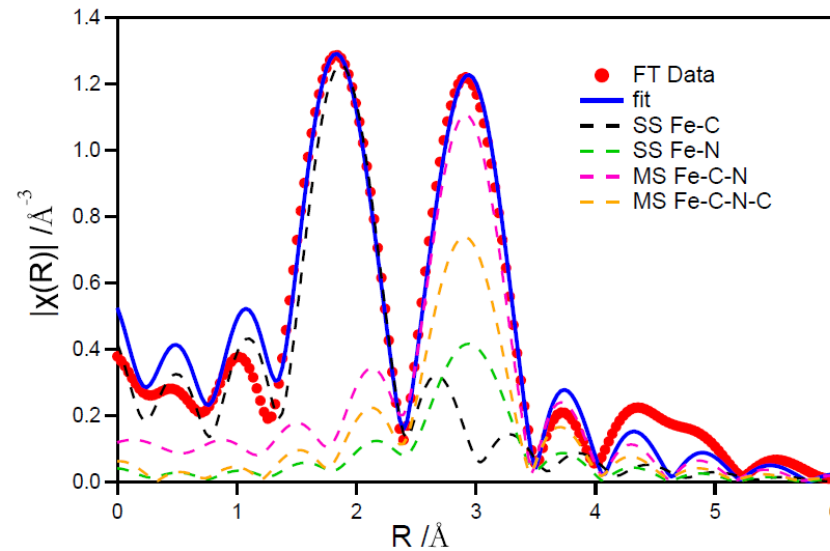
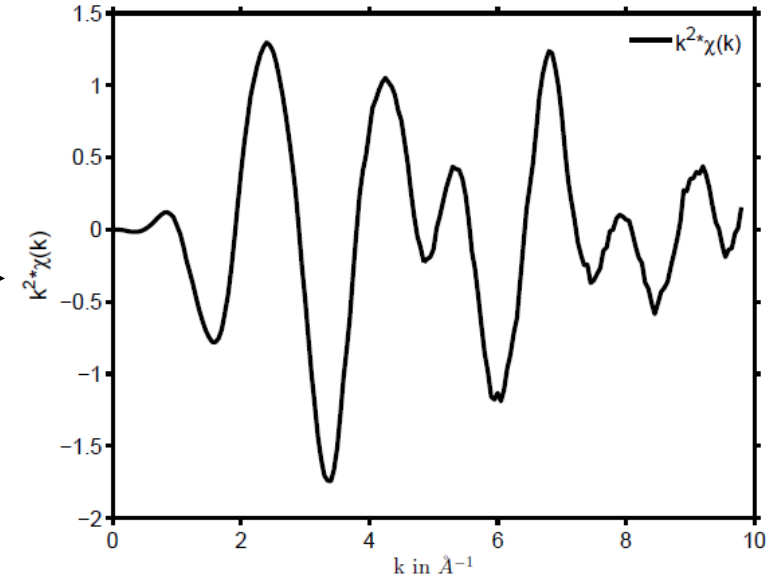
Parameters to handle:

- $R_i$  distance between the absorber and scatterer (in case of multiple scattering, half of the total travelling path)
- $N_i$  number of scattering atoms in the  $i^{\text{th}}$  shell (=coordination number)
- $F_i$  effective backscattering amplitude
- $S_0^2$  amplitude reduction factor
- $\lambda$  mean free path of the photoelectron (the probability of returning before the core hole fills up or the photoelectron scatters inelastically)
- $\sigma_i^2$  Debye-Waller factor
- $\varphi_i$  effective scattering phase shift

# EXAFS example: $[\text{Fe}^{\text{II}}(\text{CN})_6]^{4-}$



data reduction



Fourier transformation  
and fit



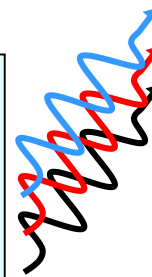
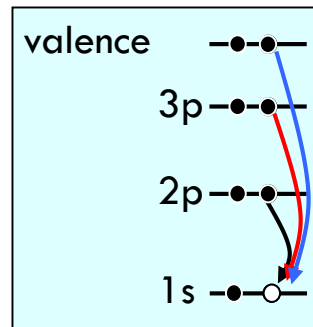
# 3. X-ray Emission Spectroscopies

## Literature:

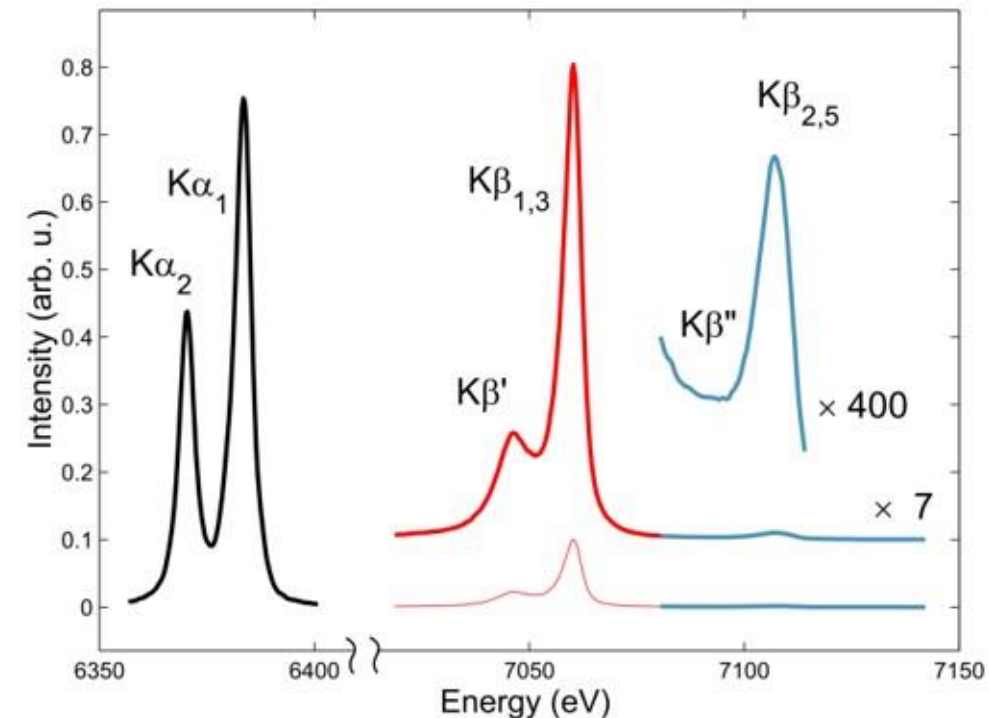
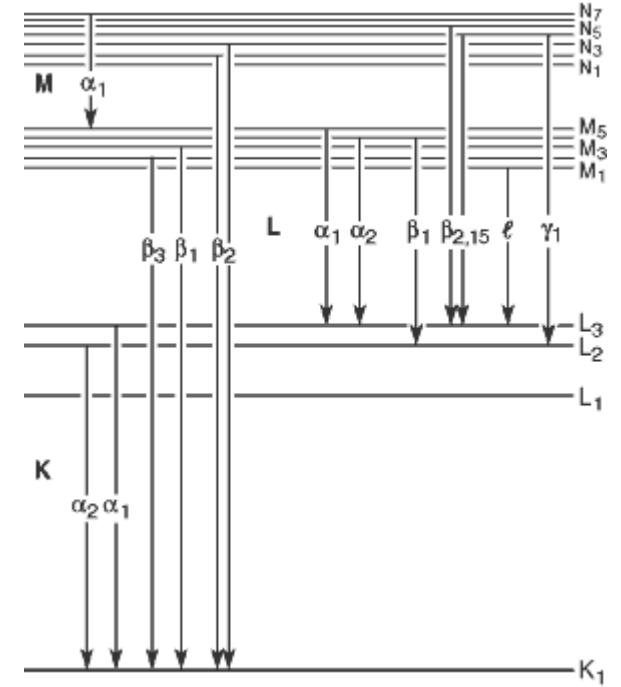
- F. de Groot: High-Resolution X-ray Emission and X-ray Absorption Spectroscopy, *Chem. Rev.* 2001, 101, 1779-1808
- P. Glatzel, U. Bergmann: High resolution 1s core hole X-ray spectroscopy in 3d transition metal complexes-electronic and structural information, *Coordination Chemistry Reviews* 249 (2005) 65–95

# X-ray emission

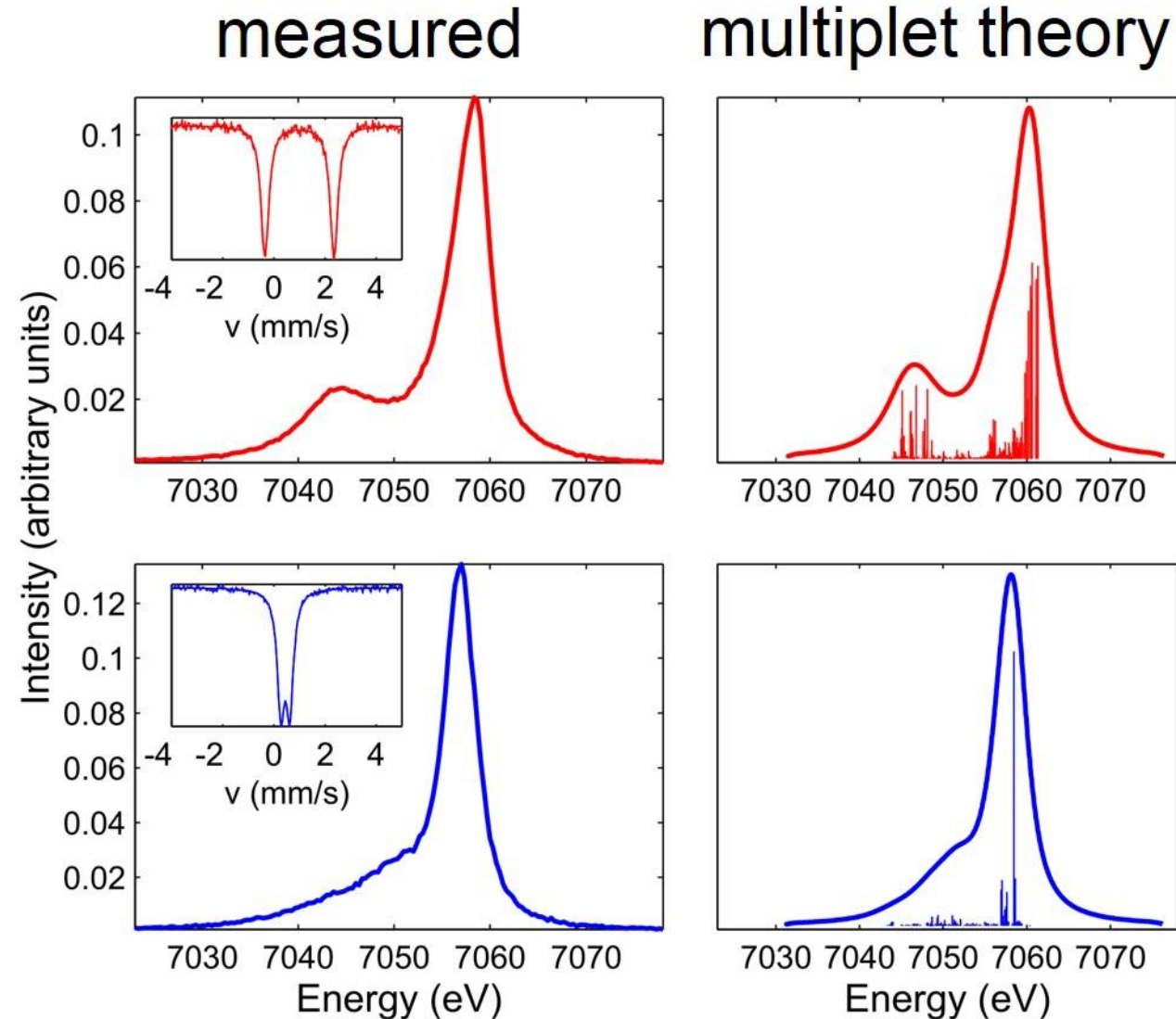
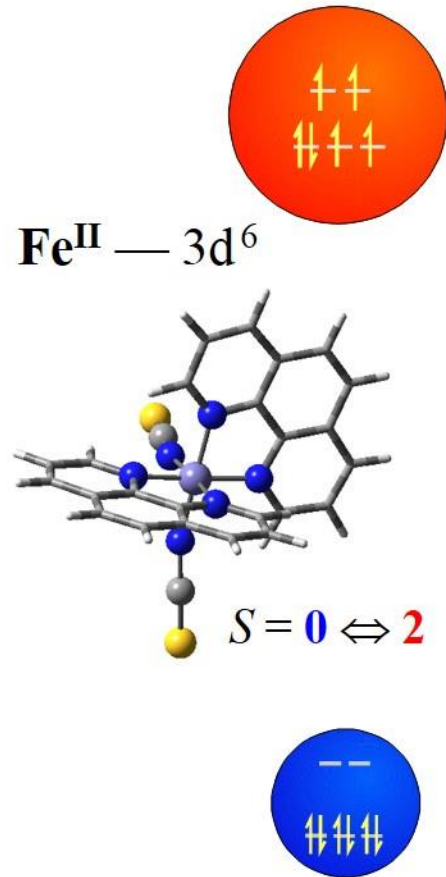
- Photon-in photon-out process: after creating the core hole, an electron from outer orbitals will fill the hole, the energy difference will be (sometimes) emitted by an X-ray photon
- Core-to-core features ( $1s2p$ ,  $1s3p$ ) sensitive mostly to the spin density via electron-electron exchange interaction, which is bigger on  $K\beta$
- Spin-orbit coupling is stronger on  $K\alpha_1$  and  $K\alpha_2$  lines (associated with  $2p_{3/2}$  and  $2p_{1/2}$ , respectively, typically 15-20 eV), but weaker on  $K\beta$  (ca. 1-2 eV)



Nomenclature:

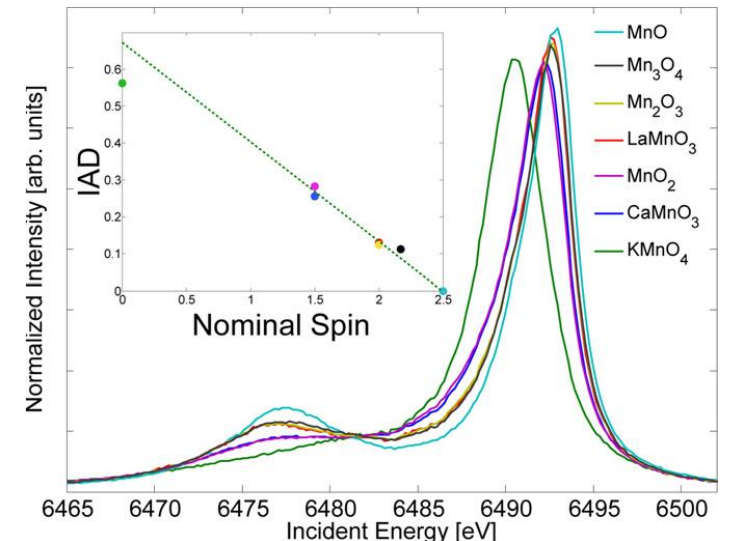
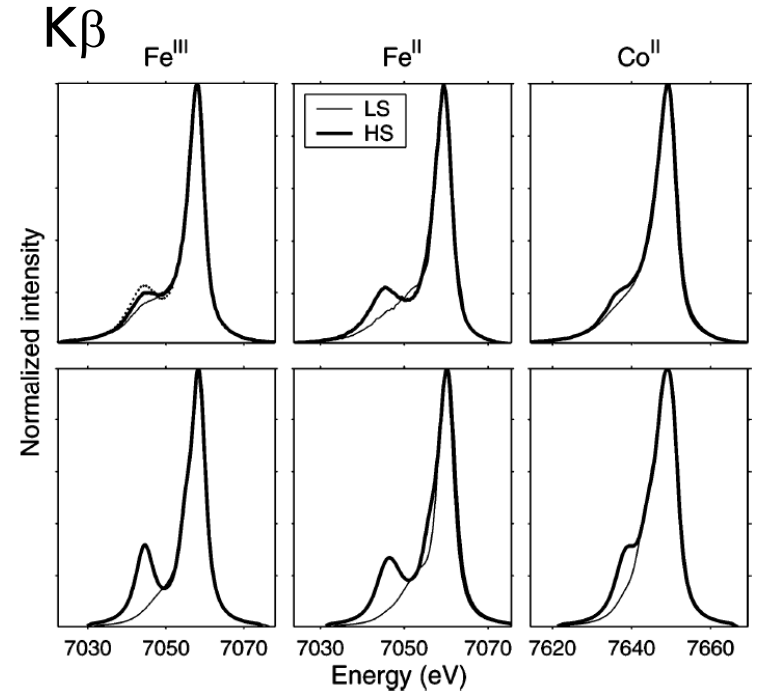
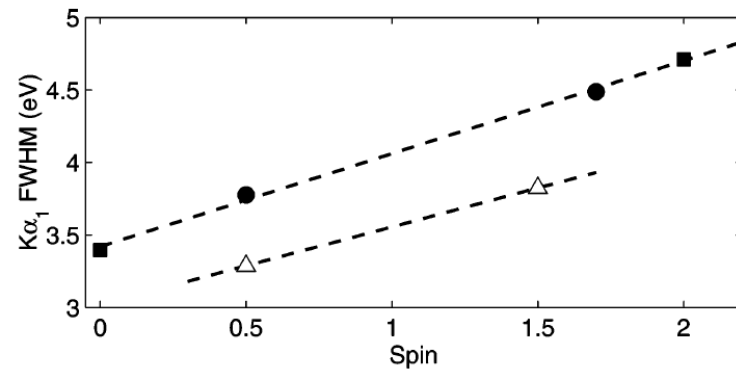
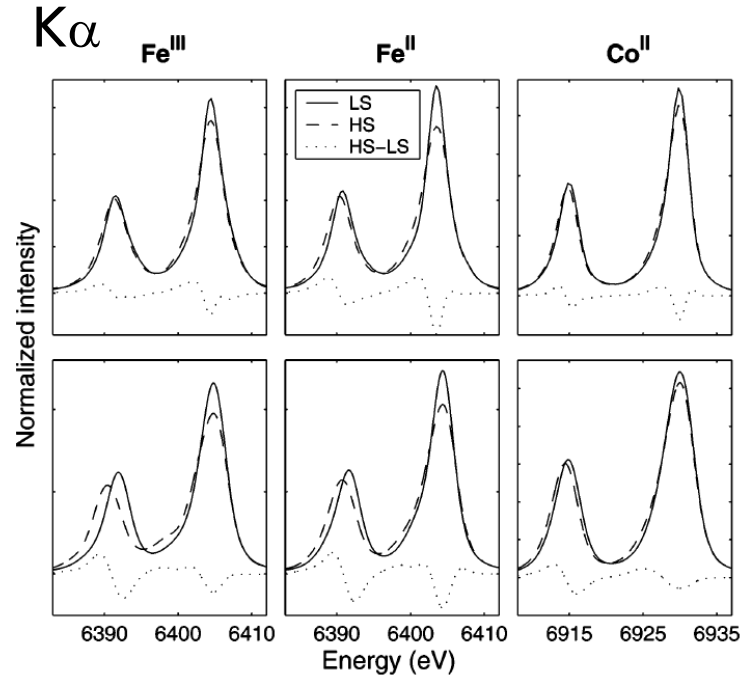


# Sensitivity of ctc XES

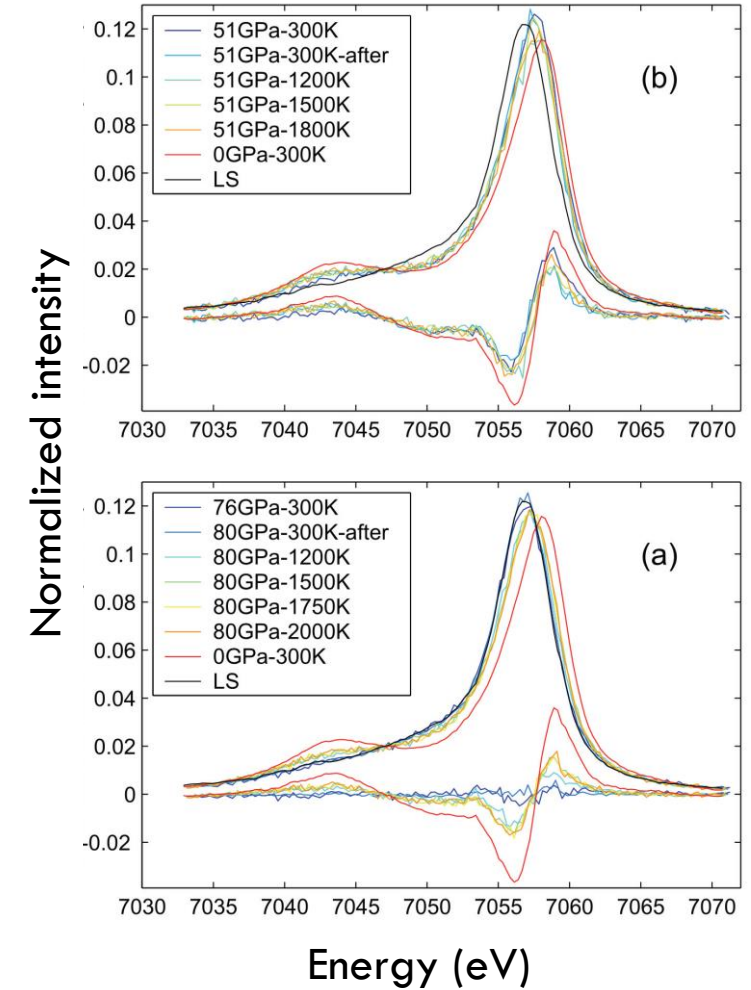
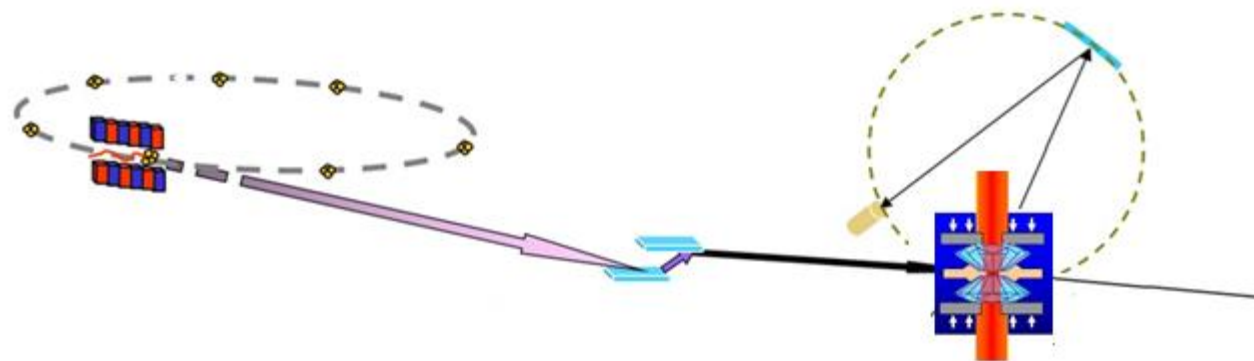
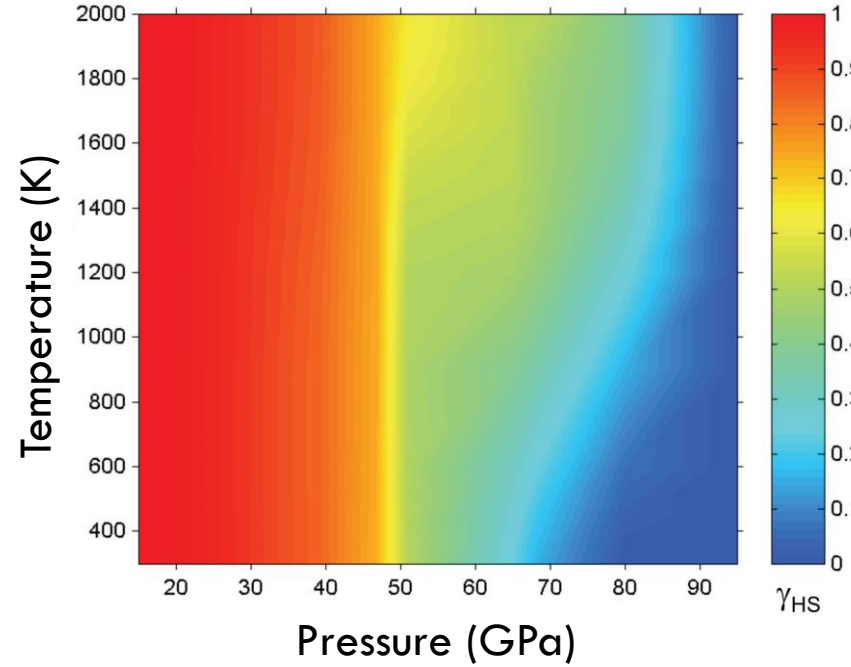
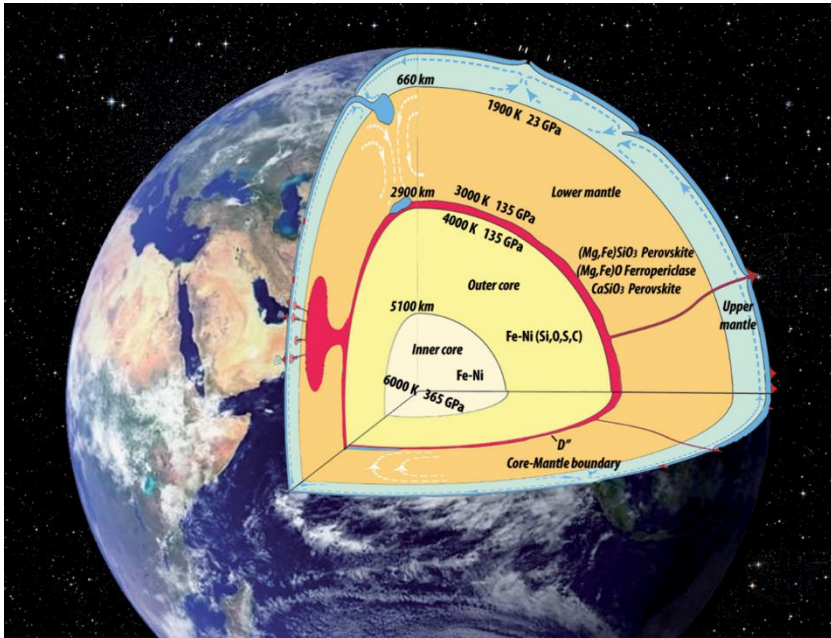


# Evaluating XES spectra

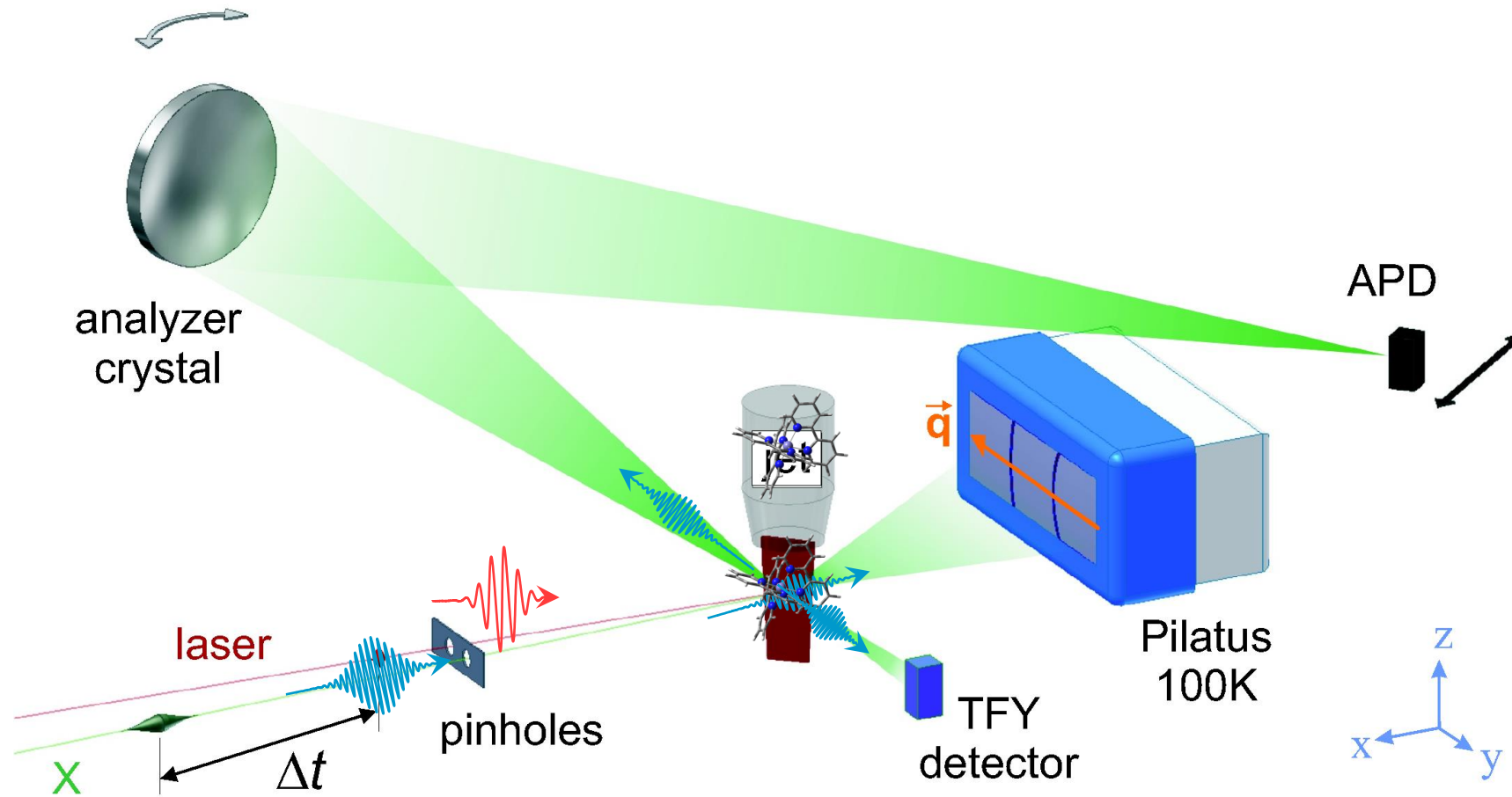
- After rigorous normalization peak widths can be used to track spin state
- IAD: integrated absolute difference, footprint compared to references



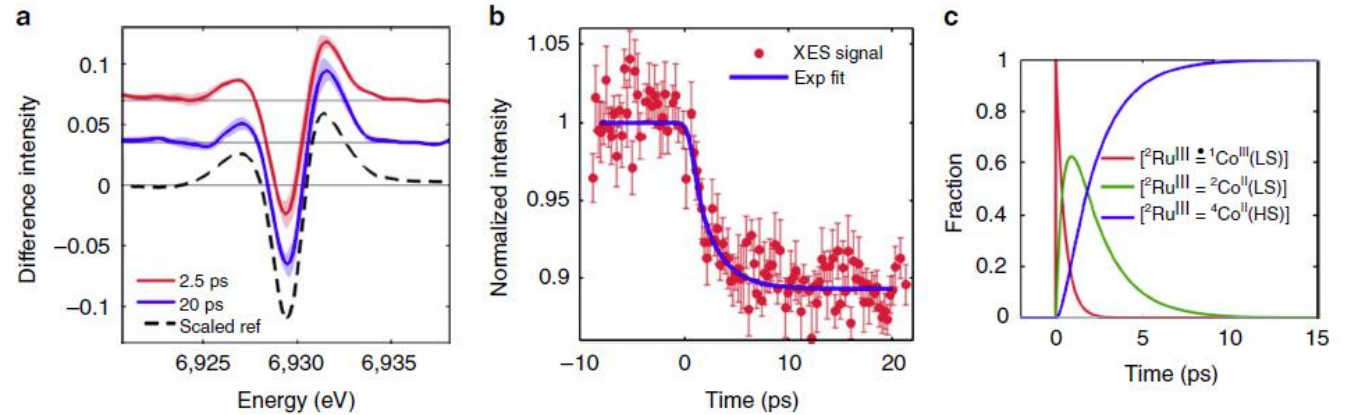
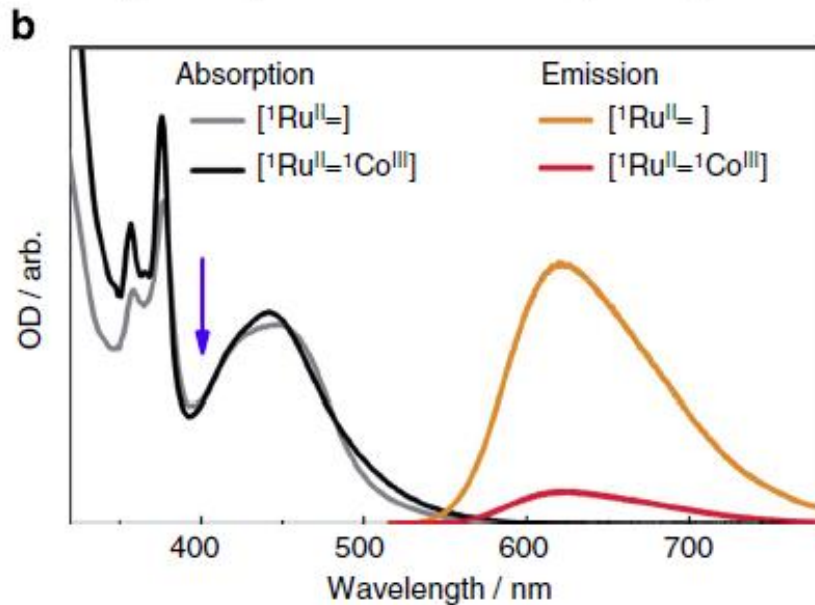
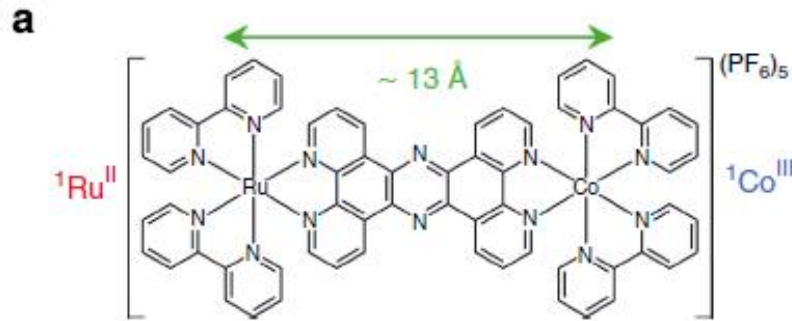
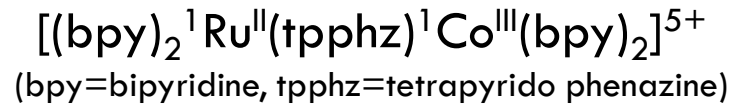
# XES example: Earth's mantle



# XES example: ultrafast processes

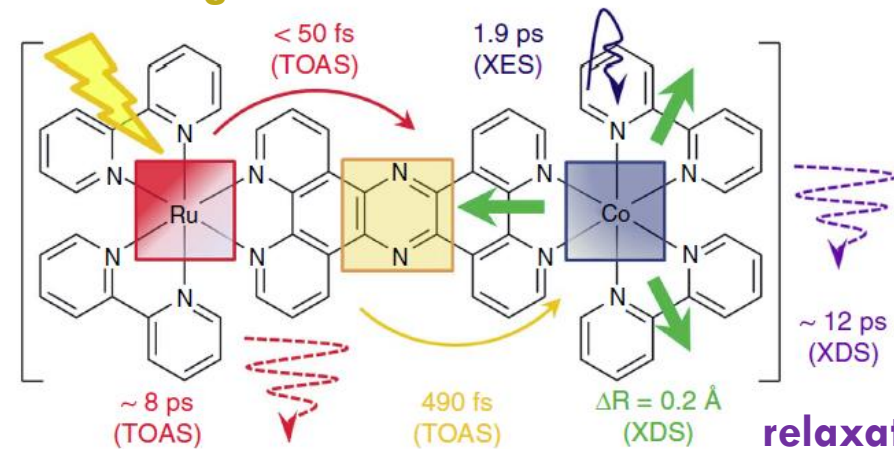


# XES example: ultrafast processes



electron leaves the Ru site,  
 transfers through the bridge,  
 leaves the bridge

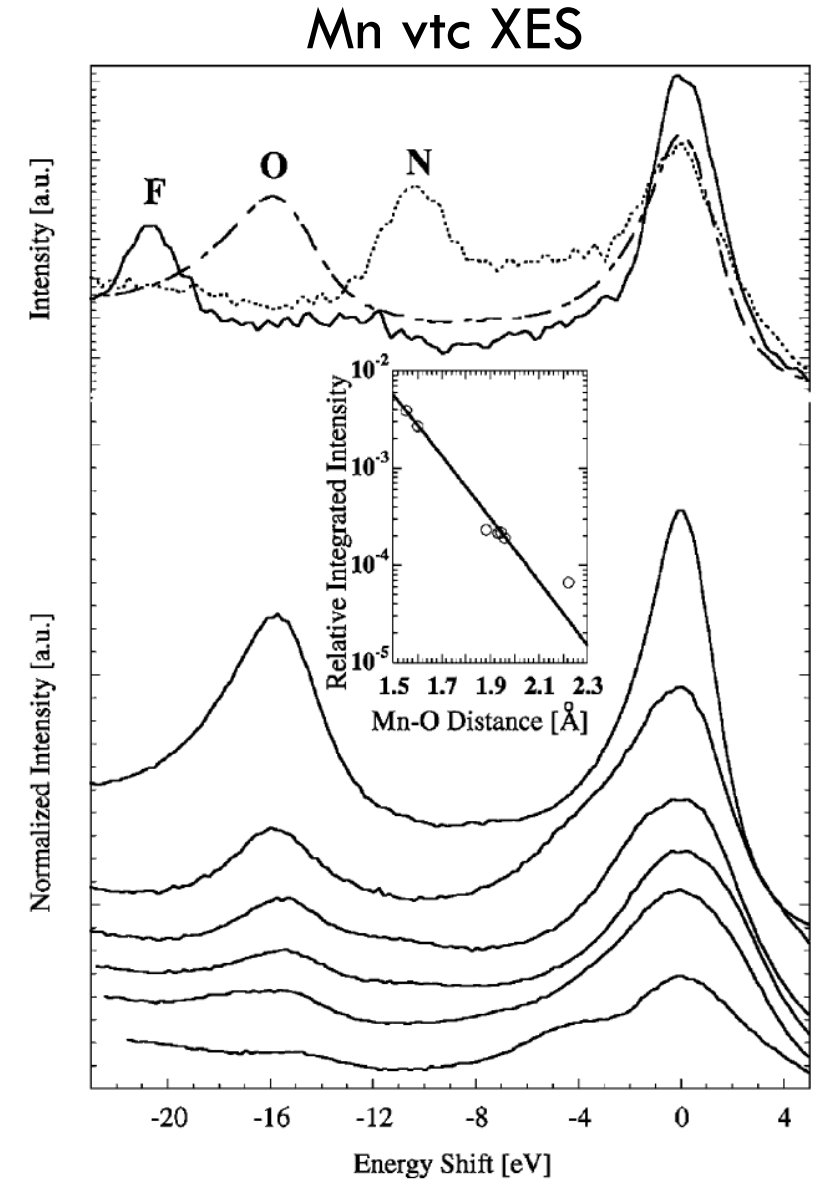
electron transfer,  
 spin-flip



relaxation and energy  
 release into the solvent

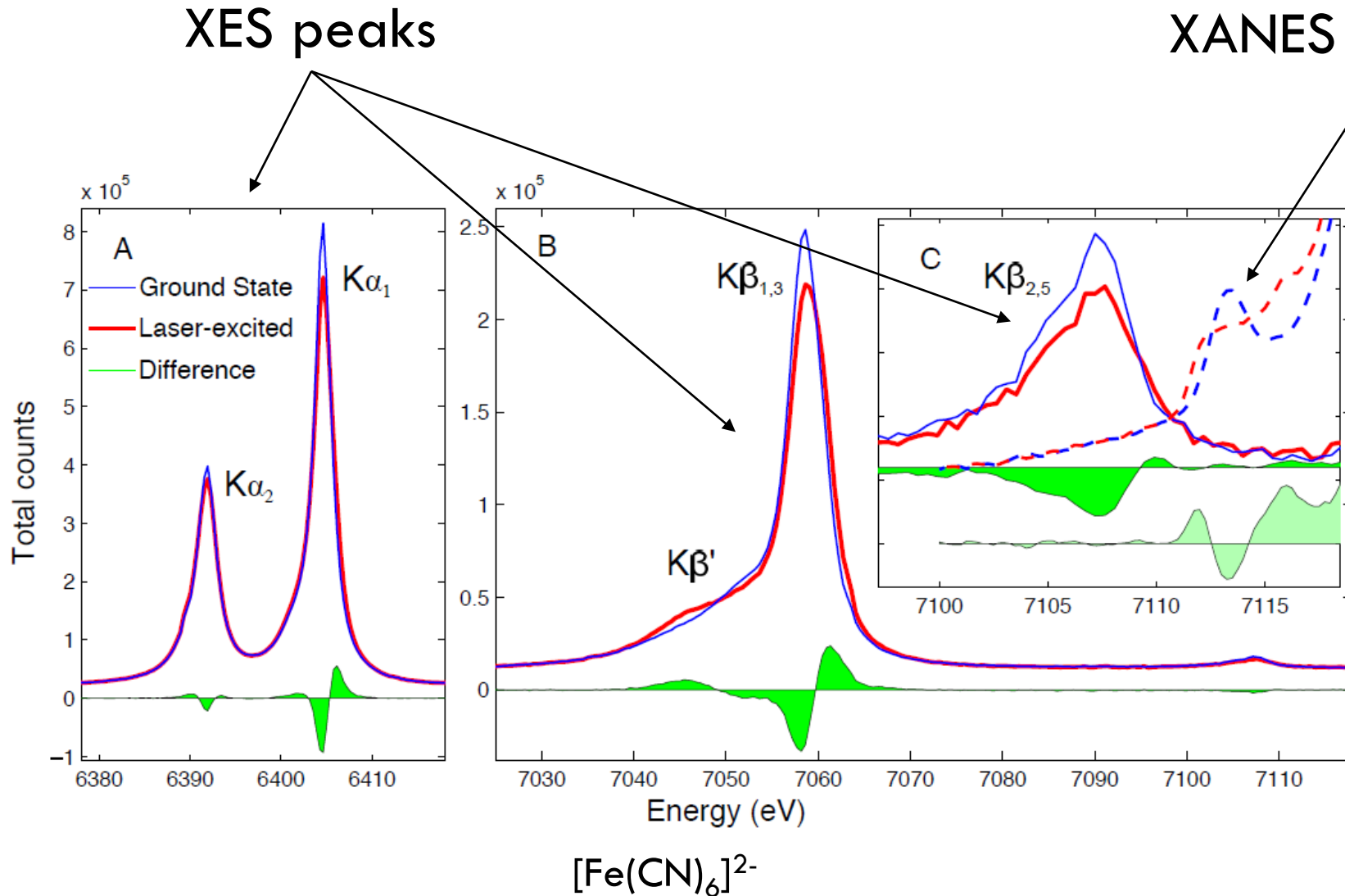
# Valence-to-core XES

- valence orbitals: chemistry!
- vtc XES reflects ligand orbitals mixed with the core ion's orbitals
- XPS has the same final state but shows also the 3d orbitals of the core ion (different selection rules!)
- vtc XES is bulk sensitive





# From XES to XAS



- XES probes the highest occupied states
- XAS probes the lowest unoccupied states
- Difference would be the valence gap, but XES has a different electronic core structure (core hole!)

# Combining XAS and XES

- $1s2p_{3/4}$  resonant XES spectra
- Scanning both the incident and fluorescent X-ray energy
- Spectral features separate better

- High Energy Resolution Fluorescent detection XANES: scanning XAS via a specific fluorescent energy
- Overcoming the  $1s$  core hole broadening

